



Materials under irradiation

PWR Structural Materials

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Short bio

Engineering degree Ecole Centrale Paris (materials major)

PhD in Applied Physics California Institute of Technology (USA)

Research assistant at Los Alamos National Laboratory (USA)

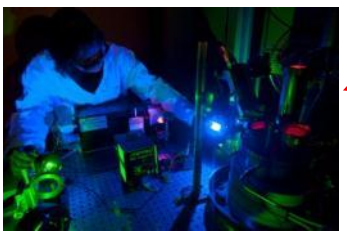
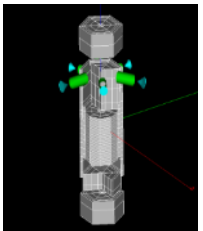
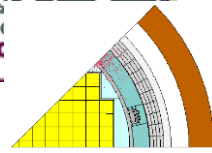
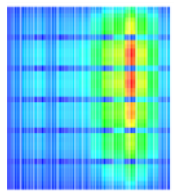
Post doc at the EU Joint Research Centre of Petten (The Netherlands)

I joined CEA in 1996, held several management positions before becoming director of nuclear activities in Saclay in 2017

Adjunct professor at Centrale Supélec, ENSTA and Phelma Grenoble



Direction of Nuclear Activities of Saclay

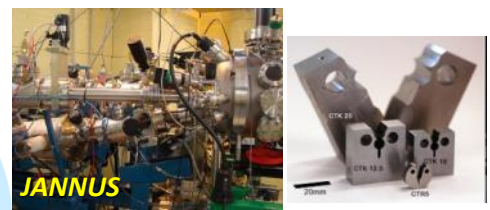
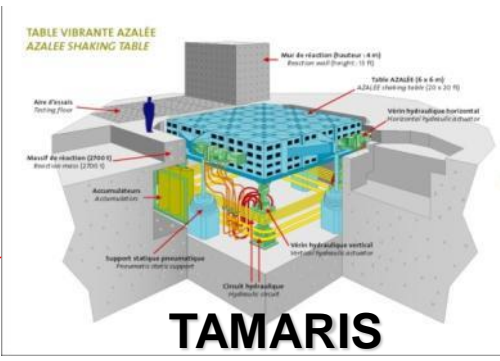


Reactor
Physics

Mechanics
and
structures

Chemistry
& analytical
sciences

Materials



Basic Sciences for nuclear applications

125 PhD students



Outline

- **Effects of irradiation on materials**
- **PWR structural materials**
 - The reactor pressure vessel
 - The internal structures
- **PWR fuel assembly materials (excluding fuel)**
- **Gen IV structural materials**



Outline

Effects of irradiation on materials

- Macroscopic effects of irradiation
- Neutron – matter interaction
- Point defects, displacement cascades, irradiation damage
- Long term evolution of point defects : structure, mobility, clusters and sinks
- Microscopic evolution

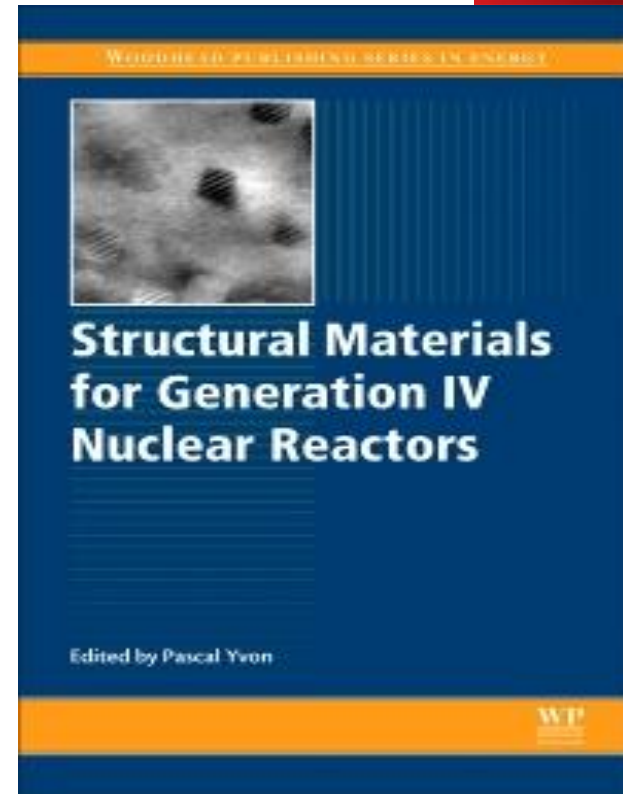
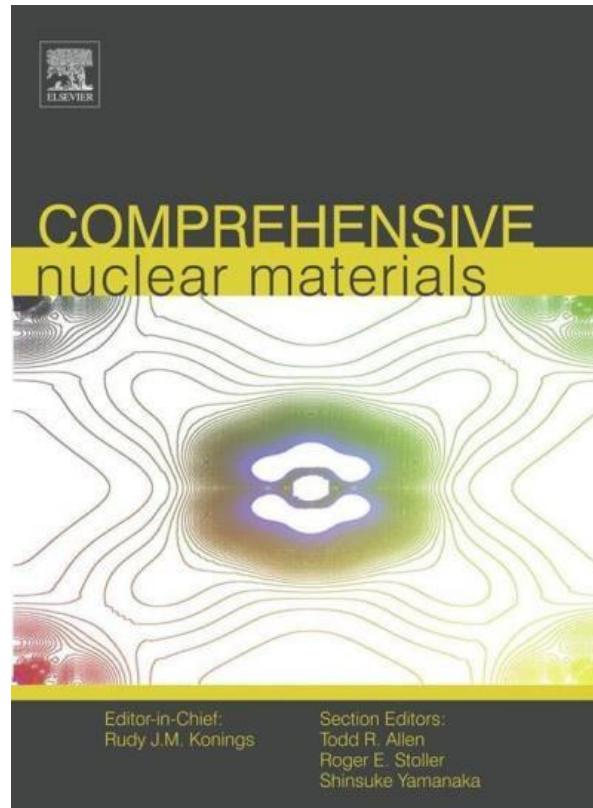
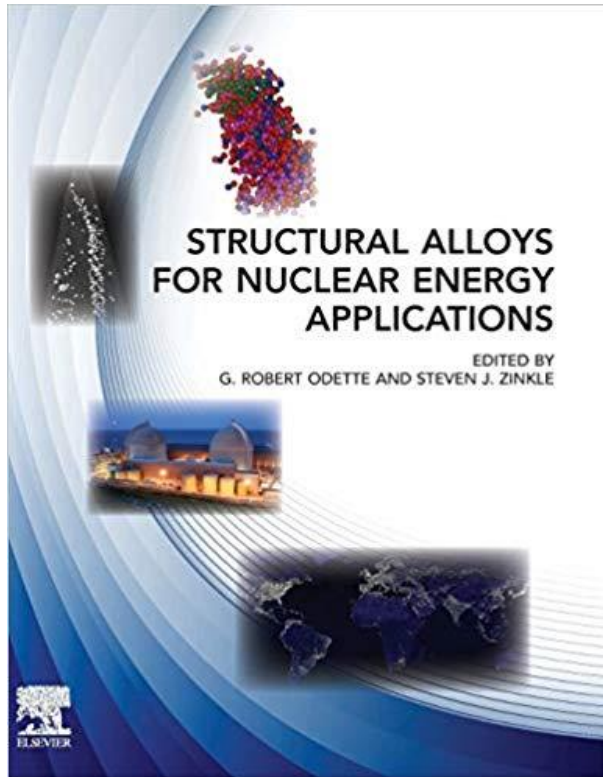
The reactor pressure vessel

- PWR design
- PWR fabrication
- Internal cladding
- Fracture of ferritic steels
- Irradiation embrittlement
- Pressure vessel integrity assessment
- Surveillance program

The internal structures

- Internals
- IASCC Intergranular corrosion (internals)
- Swelling

For further reading...



World nuclear fleet

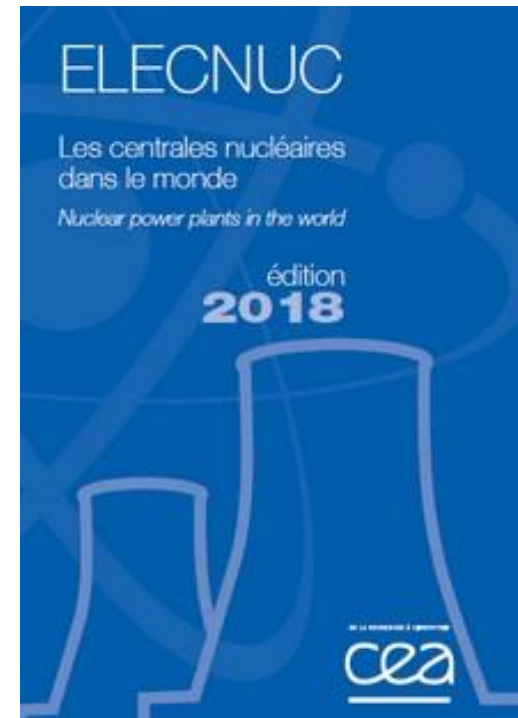


450 reactors operating in the world

>99% have a thermal spectrum

2/3 are PWR (or VVER)

So the focus will be on PWR materials



PWR Irradiated Components

Fuel Assemblies
 Zr alloys

300 - 400°C
 10/15 dpa
 5 - 6 years

Core Internals
 Nickel alloys

~ 320°C
 few 0.1 dpa
 40 → 60 years

155 bars
 293°C
 Water
 H₂, LiOH, B

~ 300°C
 0.1 dpa
 40 → 60 years

Vessel
 Bainitic steel
 16MND5
 A508 Cl 3

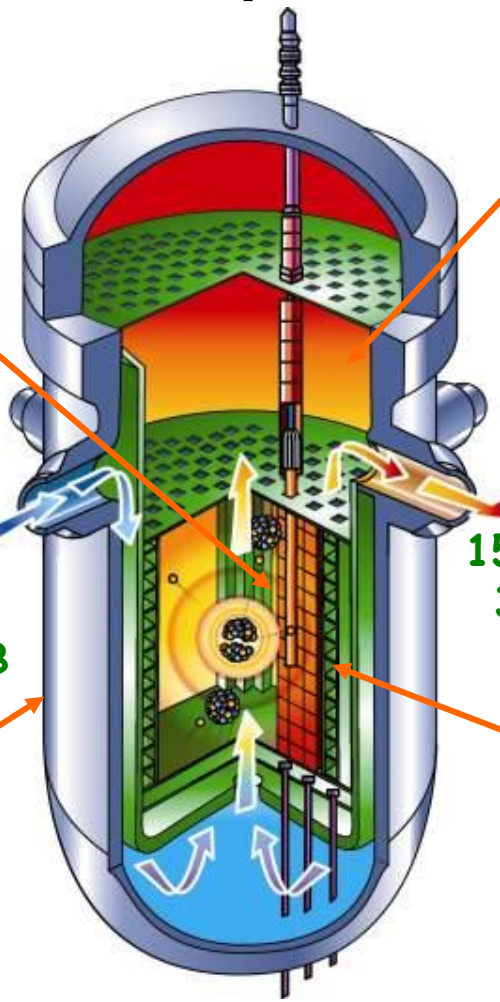
155 bars
 328°C

Control rods
 Austenitic steels

~ 320°C
 ~ 10 dpa
 few years

Core Internals
 Austenitic steels

300 - 380°C
 30 - 120 dpa
 40 → 60 years



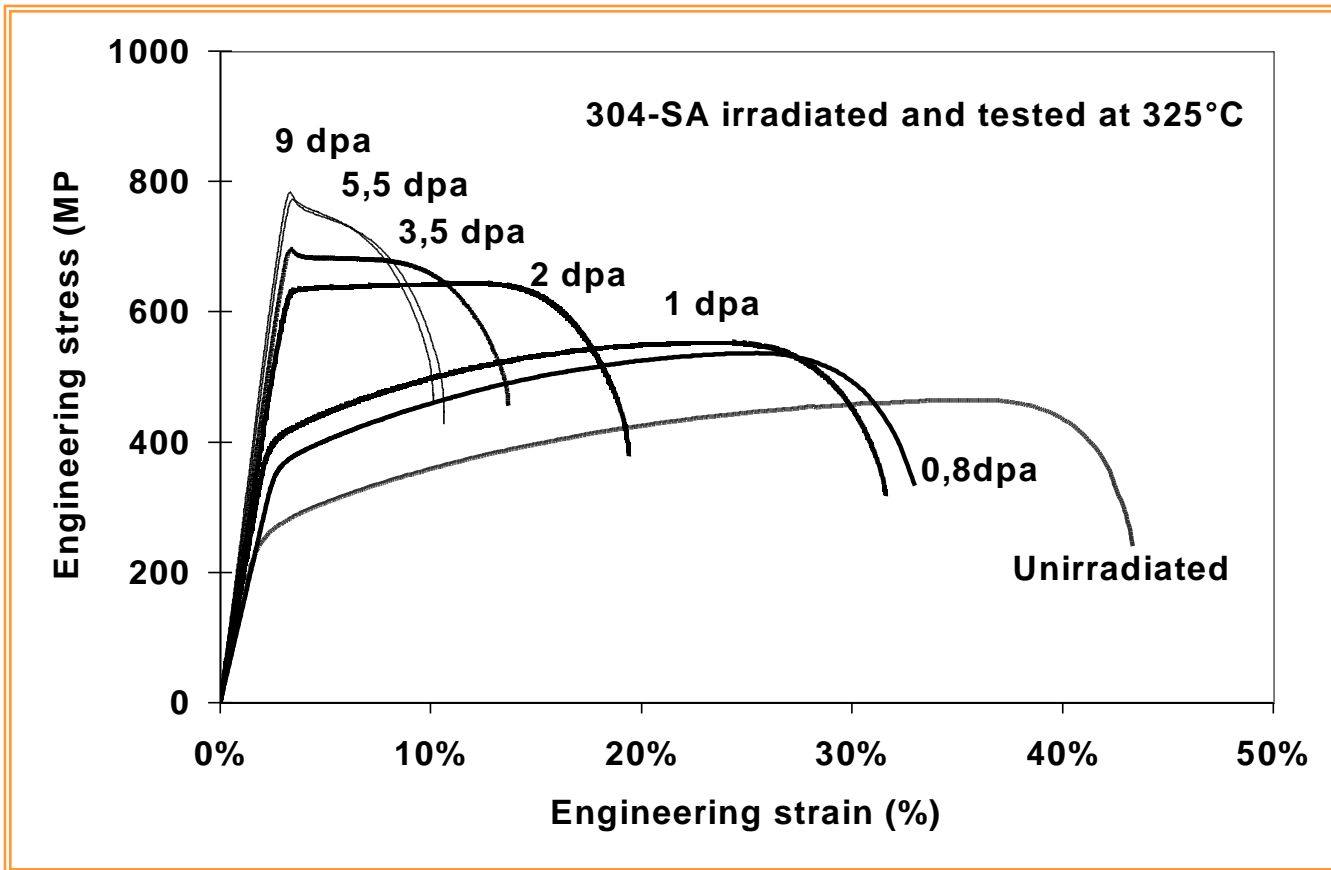
- neutrons
- temperature
- mechanical stresses
- environment

Macroscopic effects of irradiation on materials



After irradiation an evolution of mechanical properties can be observed

For instance the tensile testing properties of steel



Mechanical behavior : tensile test

Low stress

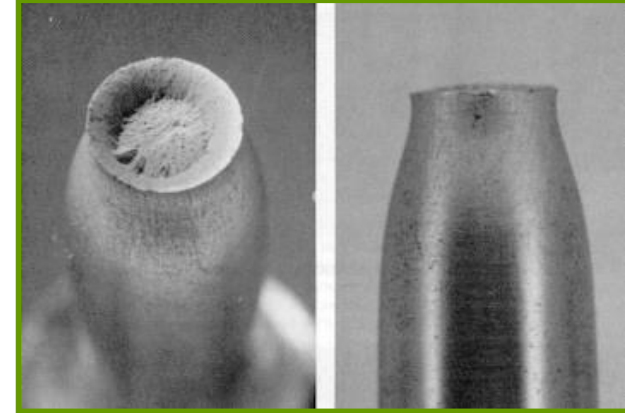
Elastic deformation (fully reversible)

$$\epsilon = \sigma / E$$

σ stress

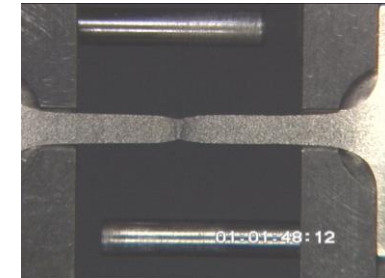
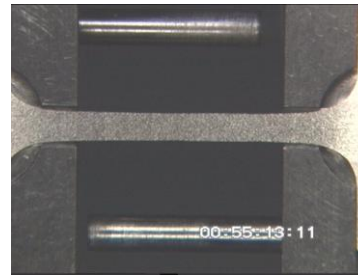
E modulus of Elasticity

(Young modulus)

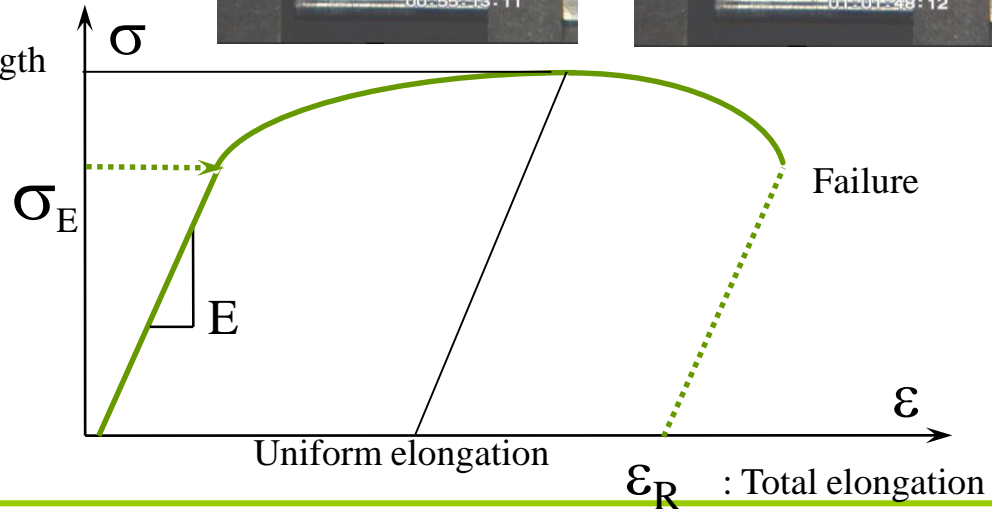


Yield strength σ_Y

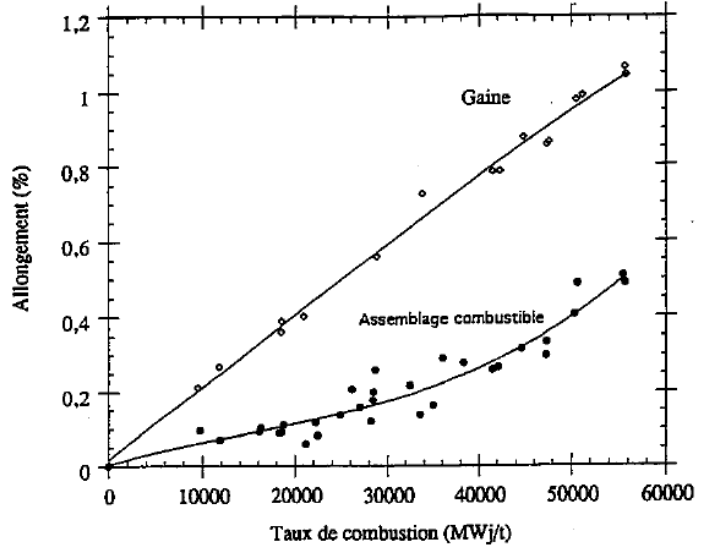
Irreversible strain induced by dislocation glide



Ultimate tensile strength

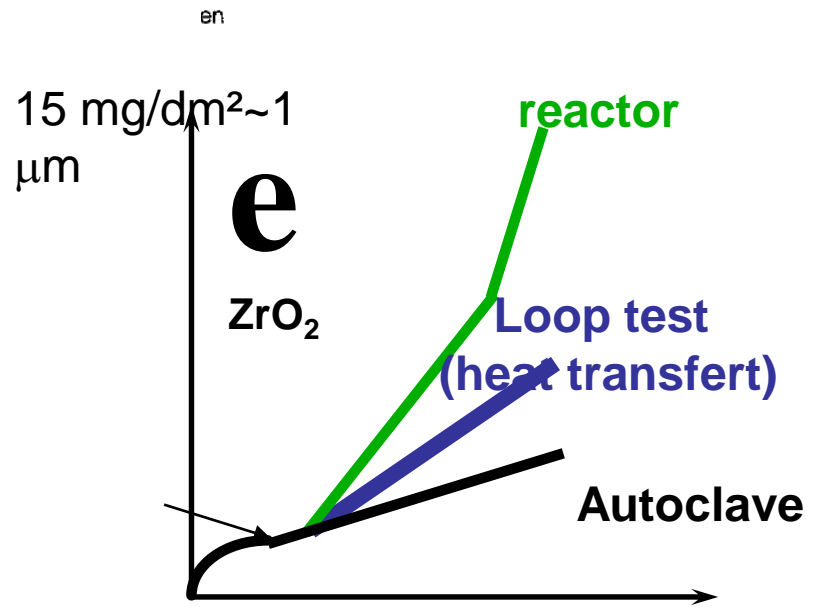
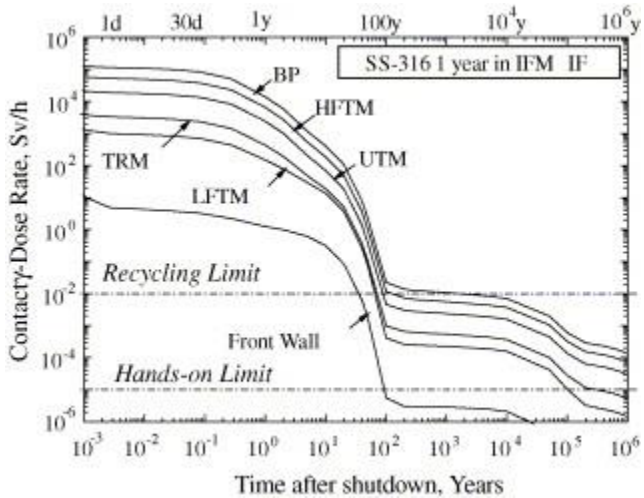


Macroscopic effects of irradiation on materials



Dimensional changes
With or without stress

Corrosion
 Creation of H₂ and He



Activation

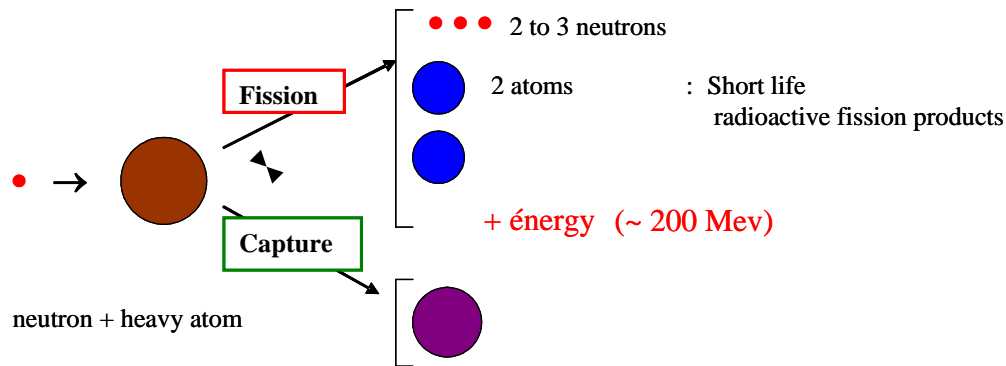
Effect of neutrons

Depending on their energies, neutron can have

Nuclear effects (inelastic): - thermal neutrons

Fission

Capture (and subsequent nuclear reactions)

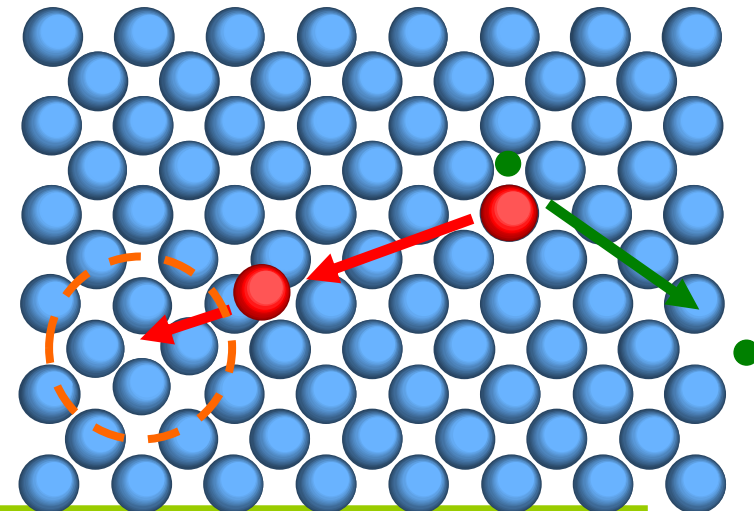


Ballistic effects (energy conservation) – fast neutrons

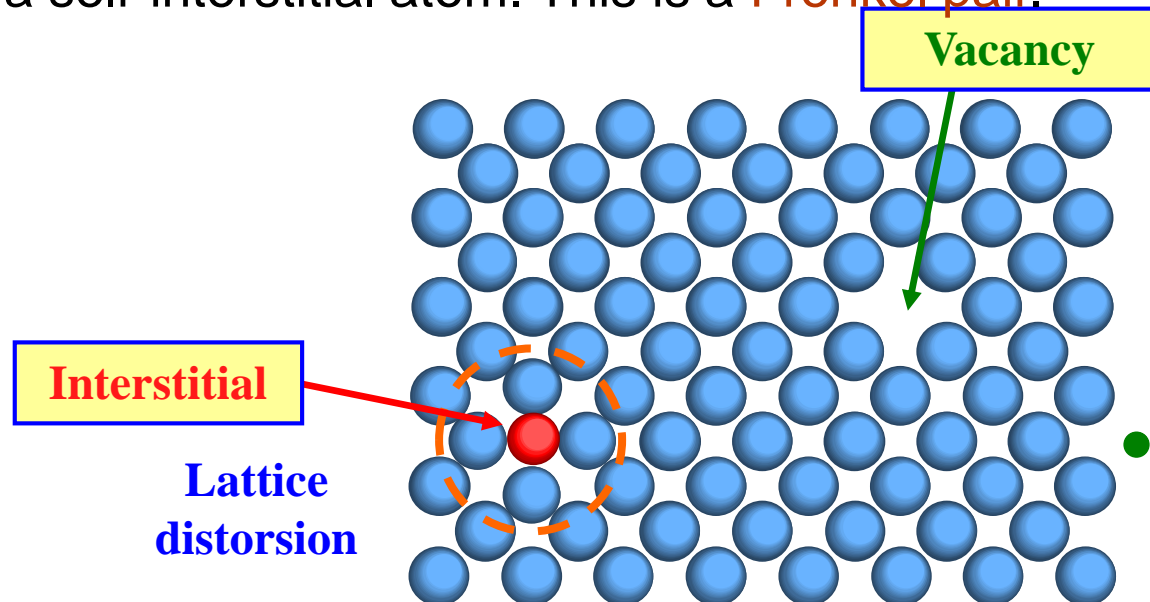
dpa – point defects

- For a transferred energy $E_t < E_d$ (threshold energy) -> vibration of the crystal lattice -> heating
- For a transferred energy $E_t > E_d$, the atom can be ejected from its atomic site and move through the crystal to other atomic sites (mean free path ~ several atomic sites)
- This creates a vacancy + a self-interstitial atom. This is a **Frenkel pair**.

**PKA : Primary
 Knock on Atom**



- For a transferred energy $E_t < E_d$ (threshold energy) -> vibration of the crystal lattice -> heating
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- This creates a vacancy + a self-interstitial atom. This is a **Frenkel pair**.



Neutron spectrum and units of irradiation



The neutron of different energies have different effects on the materials

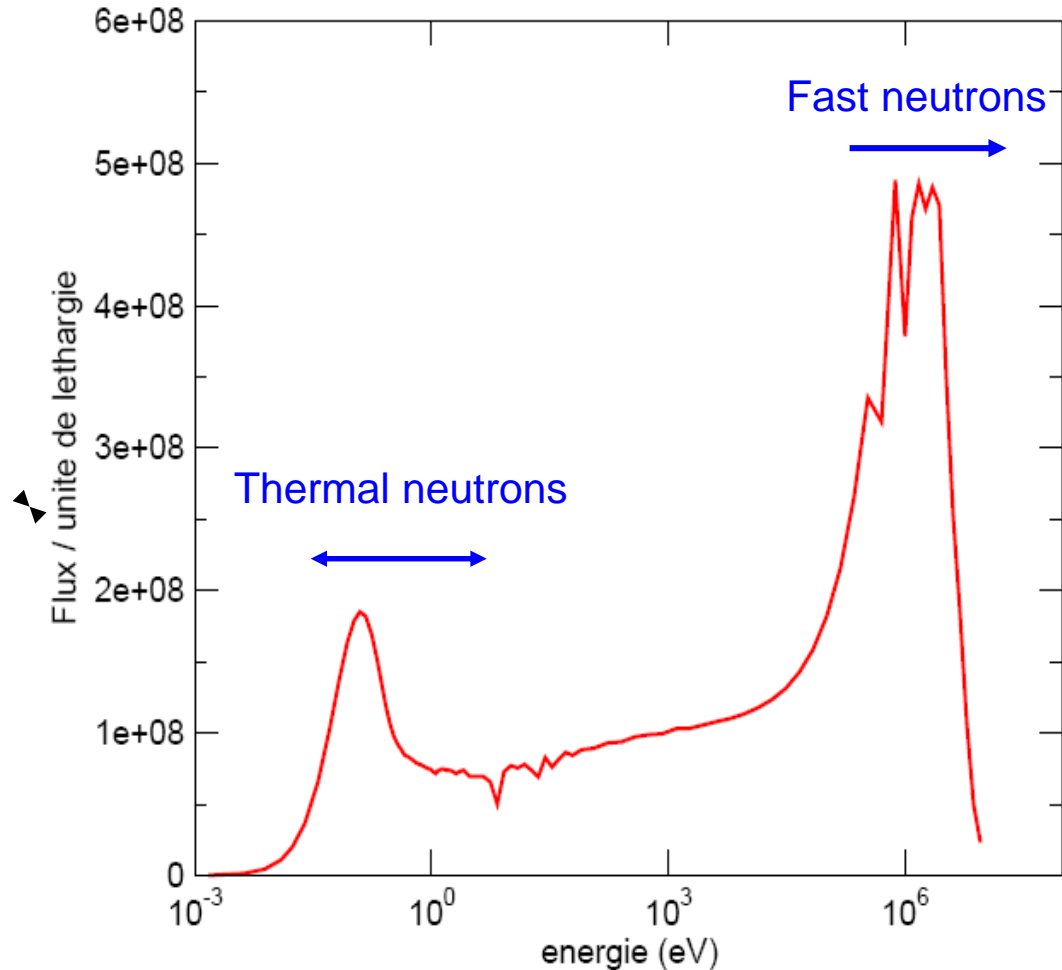
Unit of flux : $n.cm^{-2}.s^{-1}$

Unit of fluence: $n.cm^{-2}$

Difficult to describe a spectrum by just one number

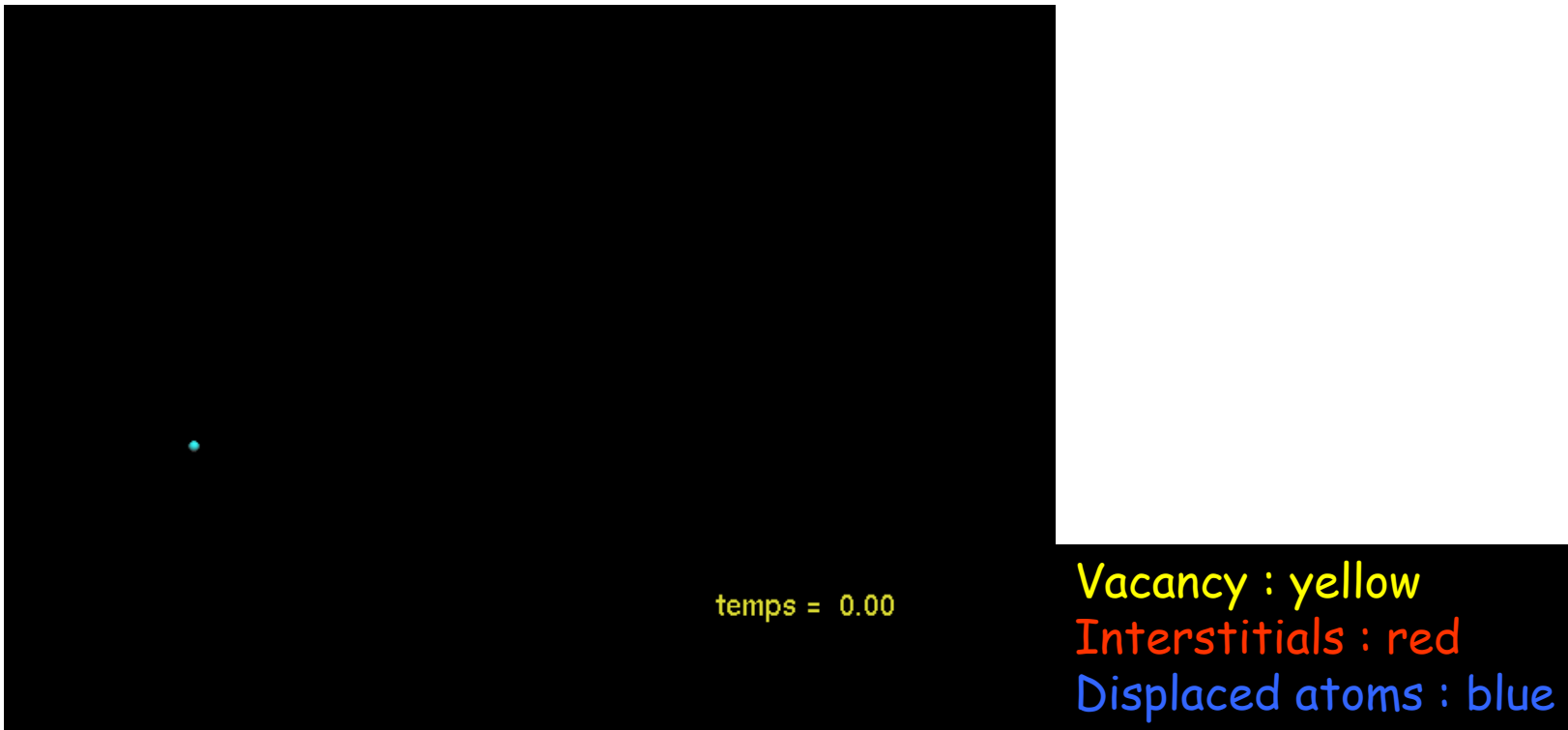
Therefore use of the dpa (displacement per atom)

PWR neutron spectrum

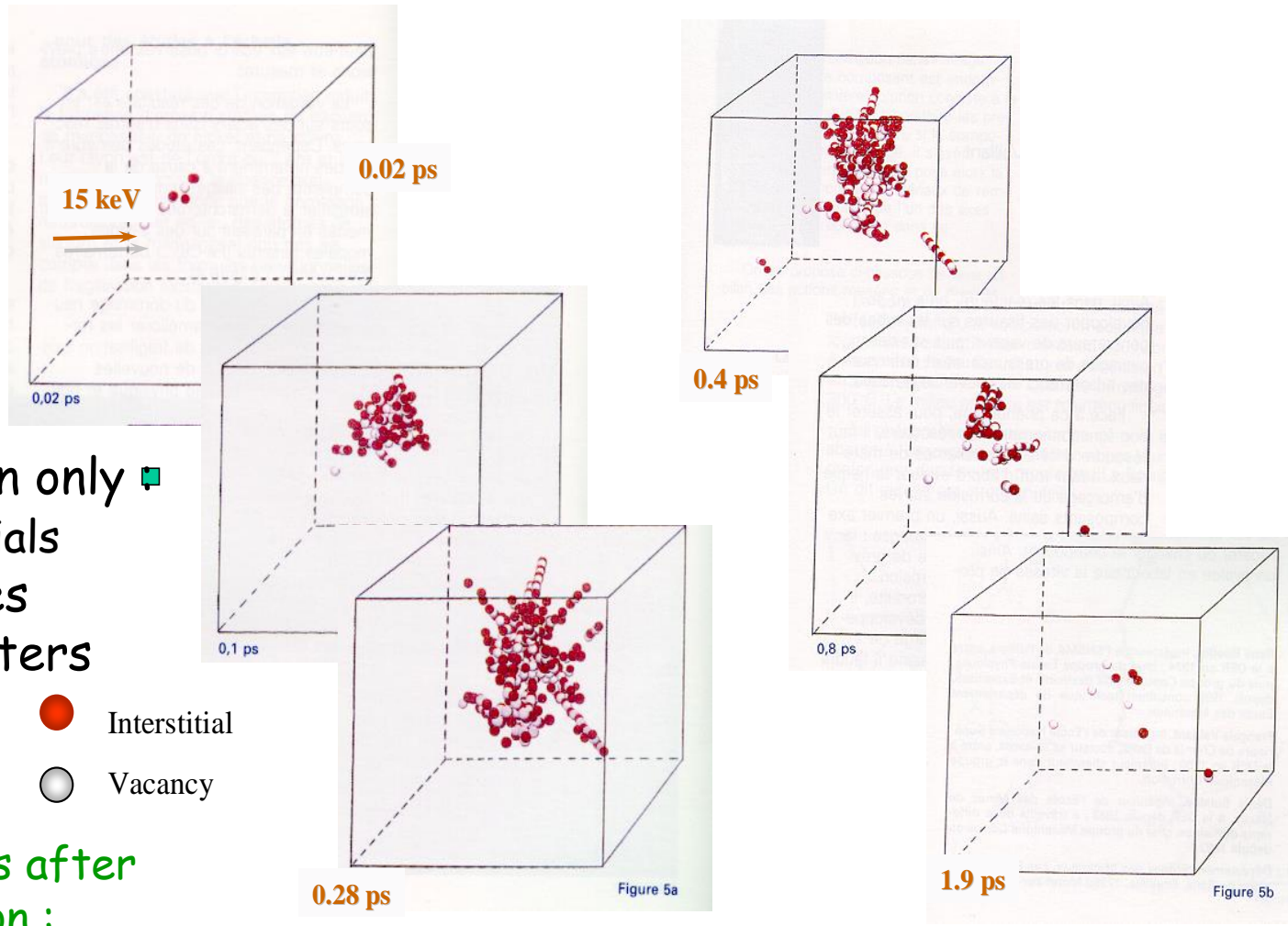


Displacement cascade ($E_t \gg E_d$)

- For a transferred energy large compared to E_d , the ejected atom transfers part of its energy to other atoms of the crystal lattice...
... these other atoms can then displace other atoms.
- The primary knock on atom induces a displacement cascade

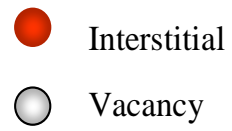


Cascade evolution



After 10 ps, remain only

- Isolated interstitials
- Isolated vacancies
- Interstitials clusters
- Vacancy clusters

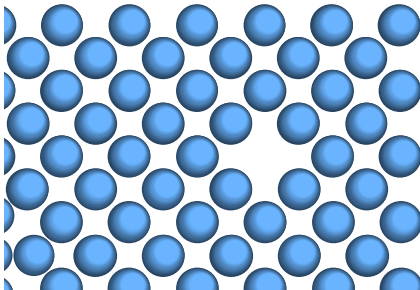


Few surviving defects after cascade relaxation : only 1/100 of displaced atoms remain

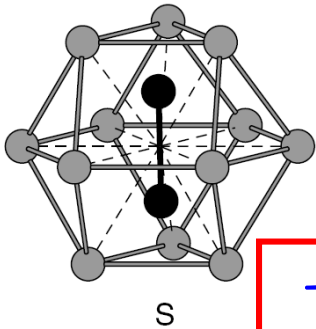
-> Cascade creation (in agreement with KP)

-> Cascade relaxation recombinations

Formation energy of point defects



Vacancy formation energy :
low distortion of the crystal lattice -> low formation energy



interstitial formation energy :
High distortion of the crystal lattice -> high formation energy

-> Out of reactor it is much easier to create a vacancy than a self-interstitial

Point defect concentration at thermodynamic equilibrium

$$C_{DPe} = \exp\left(-\frac{H_{DP}^f}{kT}\right)$$

← Formation energy

-> No interstitial at thermodynamic equilibrium

2 important phenomena

Irradiation damage : creation of point defects and mixing of the atoms (dpa)

But also

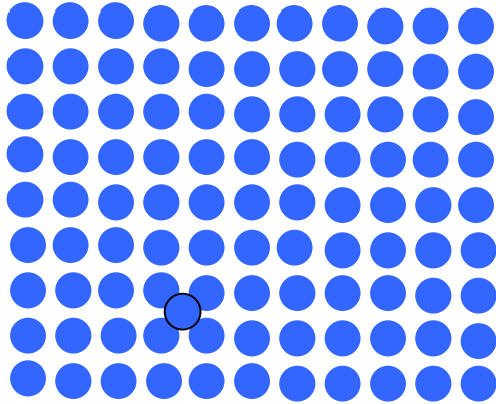
Evolution of these point defects

- Recombination
- Clustering
- Annihilation on sinks

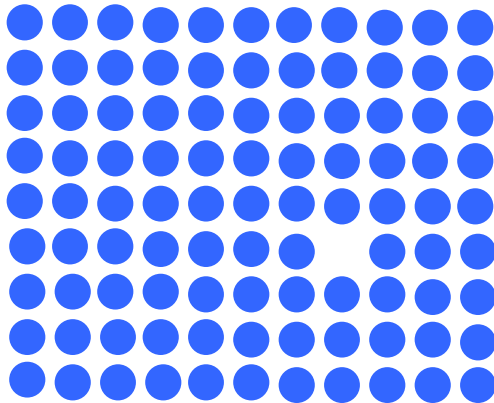


Mobility of point defects

Interstitial



Vacancy



$$D_{DP} = D_{DP0} \exp\left(-\frac{H_{DP}^m}{kT}\right)$$

	Vacancy	Interstitial
Formation energy (eV)	1.4-2.1	2.8-3.5
Migration energy (eV)	0.5-0.9	E _a =0.01-0.06 E _c =0.1-0.3

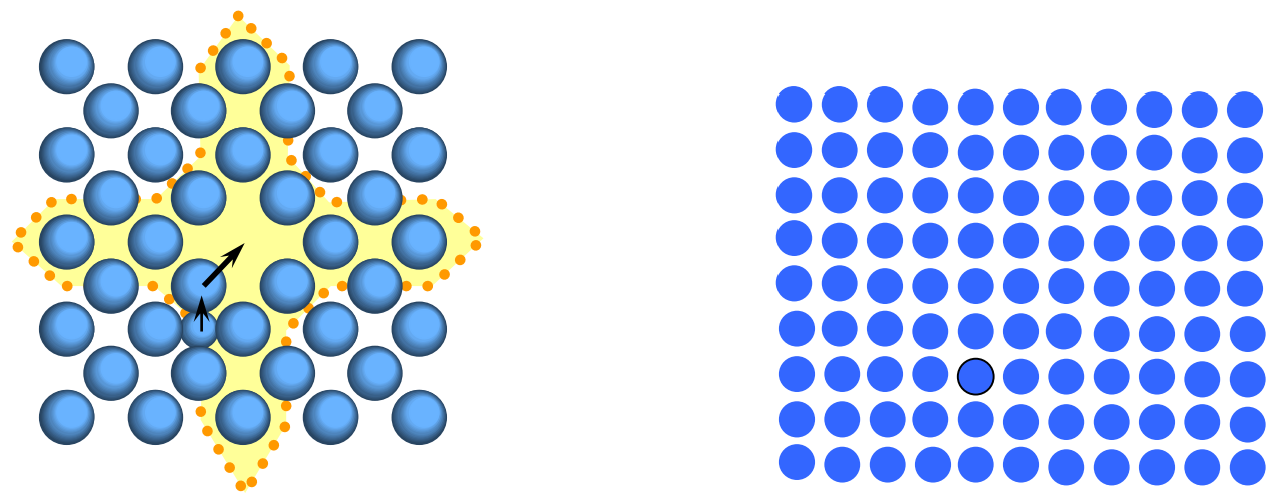
Anisotropic diffusion of the self-interstitial

-> High mobility for interstitials, low mobility for vacancies

Spontaneous mutual recombination

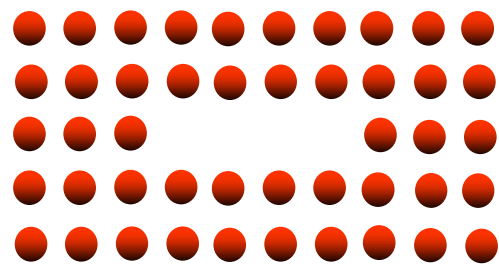
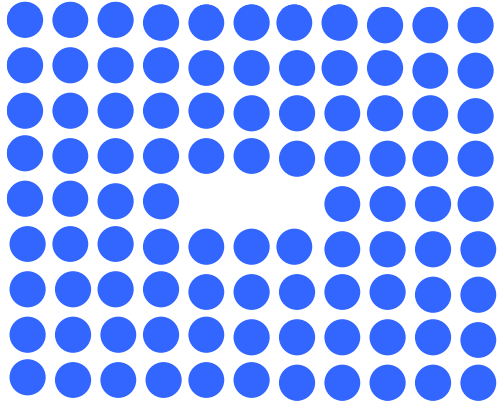
A self-interstitial close to a vacancy reacts spontaneously and the two defects are annihilated

- The mutual recombination volume is about 100 atoms.
- Saturation of the point defects (if no other sink) due to mutual recombination

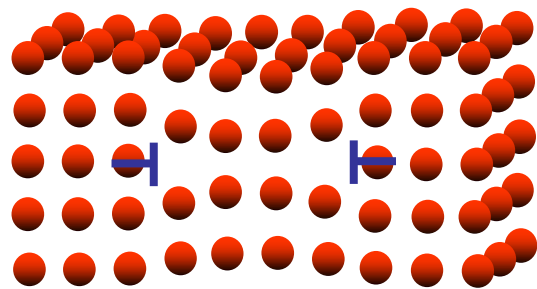


Mutual recombination :
 $\square + \circ = \text{nothing (if pure material)}$

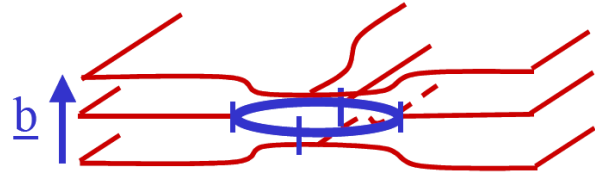
Point defect clustering



Vacancy disk



= dislocation loop

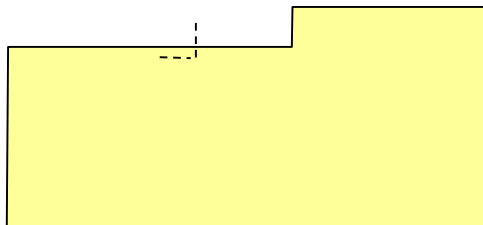


Point defects clustering : loop, cavity

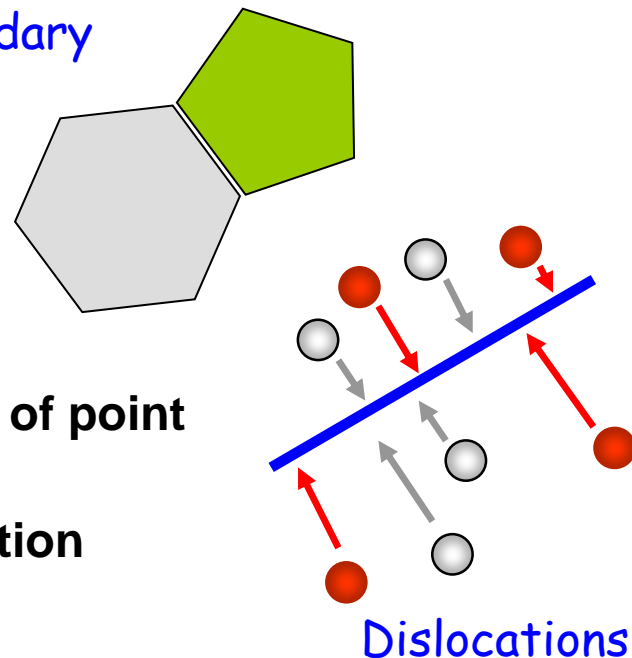
$\square + \square = \square\square + \square = \square\square\square + \square = \square\square\square\square$ or $\begin{matrix} \square & \square \\ \square & \square \end{matrix} \rightarrow$ Point defect cluster
 $\circ + \circ = \circ\circ + \circ = \circ\circ\circ + \circ = \circ\circ\circ\circ$

Annihilation on other sinks

Surface



Grain boundary



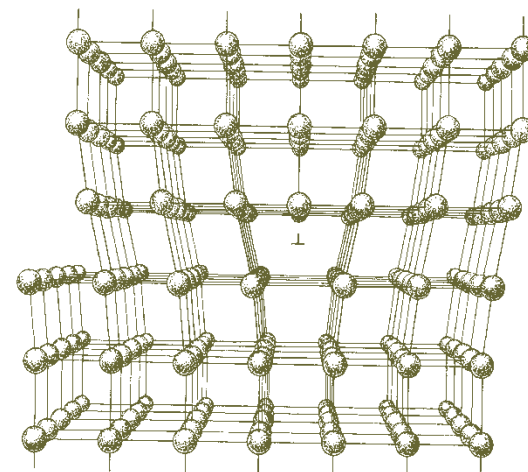
Dislocations

Close to the sink : equilibrium concentration of point defects

Sur-saturation of point defects due to irradiation

-> point defect flux towards the sinks

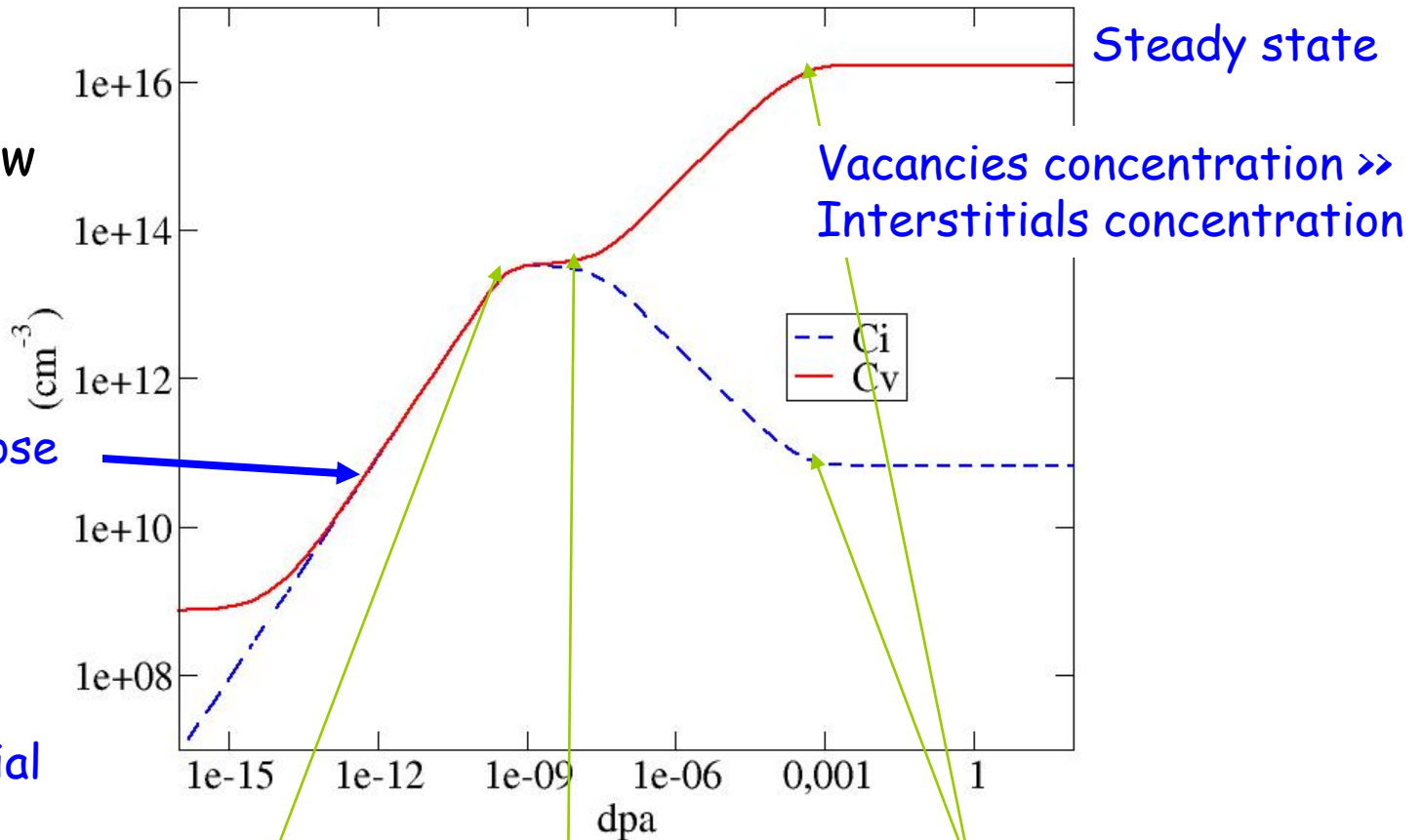
- High interaction between interstitial-dislocation
- Low interaction between vacancy-dislocation
- -> Higher chemical force on interstitials -
 > higher interstitial flux towards dislocations (if other types of sinks, such as grain boundaries)



Evolution of the point defects concentration



Iron, $2 \cdot 10^{-8}$ dpa/s, 300°C , low sink density.



Hypothesis : Low sink density

Increase with dose

Thermodynamic equilibrium vacancies

No interstitial

Start of recombinations

Interstitials reach sinks
Because of a higher mobility

Vacancies reach sinks

Effect of the irradiation temperature

- There is competition between the creation of point defects and their elimination
- At low temperature, the point defect mobility is reduced. High density of point defects. The point defects are mainly annihilated by mutual recombination.
- At high temperature, the vacancies concentration at thermodynamic equilibrium is high, the increase due to irradiation in point defects is low.

When the temperature increases, so does the mobility of defects. Therefore, the system tends to go back to the equilibrium state

Beware of accumulation of energy



Microstructural evolution



Effect of elastic collisions on precipitates

- Amorphization
- Dissolution

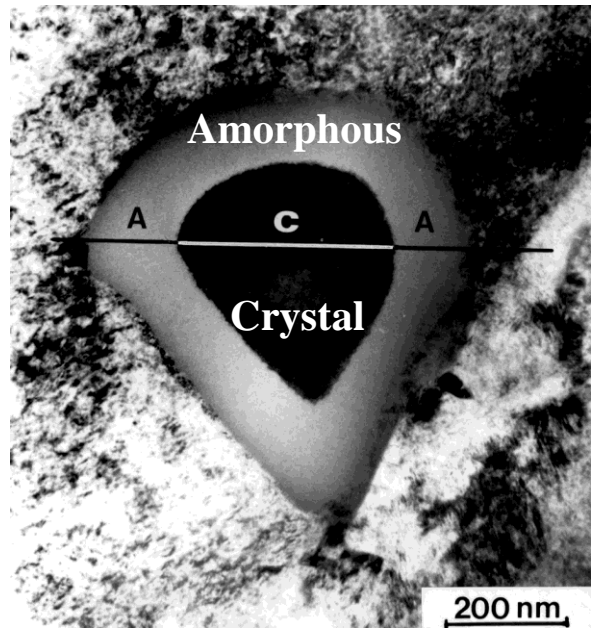
Enhanced diffusion due to the super-saturation of point defects

- Precipitation

Nuclear reactions

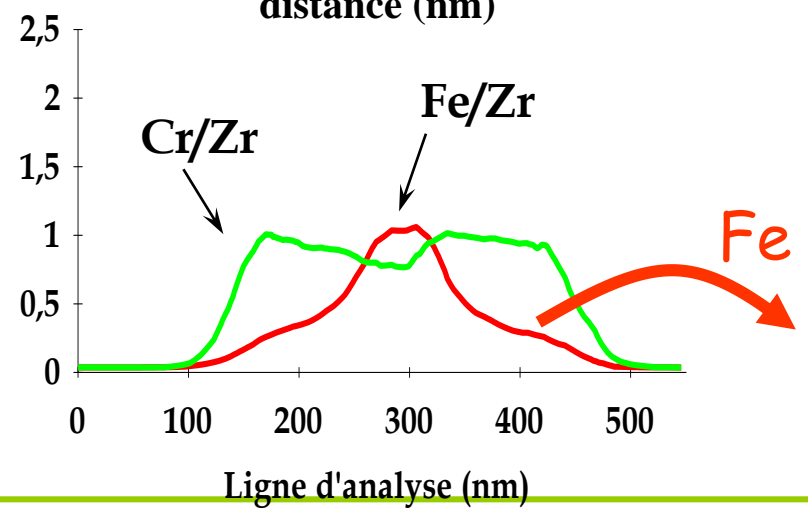
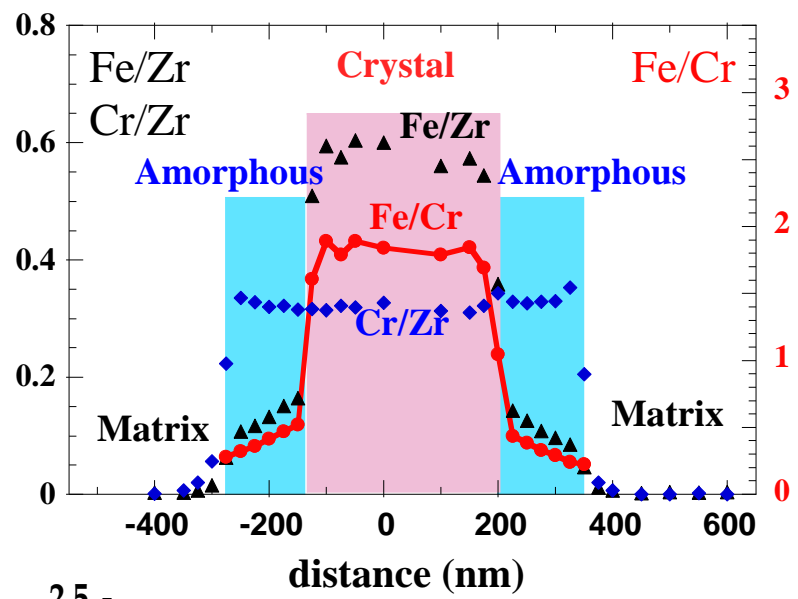
- Changes in chemical composition

Amorphization of precipitates



Laves phases
 $(Fe,Cr)_2Zr$

Amorphization of the precipitates
 Dissolution of Fe into the matrix

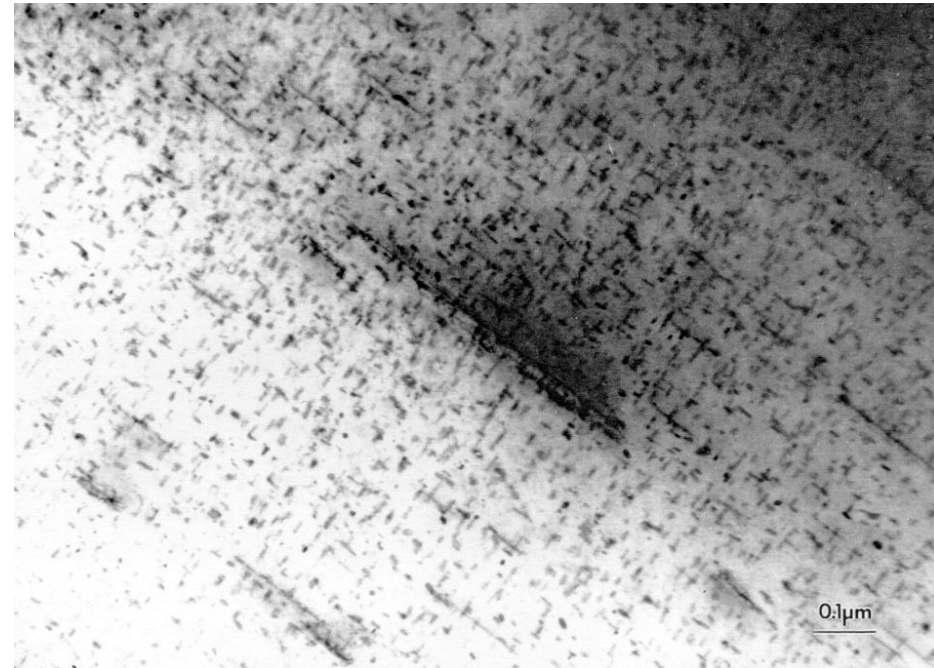




Precipitation of β Nb under irradiation in Zr alloys with Nb

Zr-Nb alloys, solide solution super-saturated in Nb (out of thermodynamic equilibrium).

Under irradiation : the high vacancies concentration leads to a high vacancy flux that enable a fast return to thermodynamic equilibrium, and re-precipitation of β -Nb

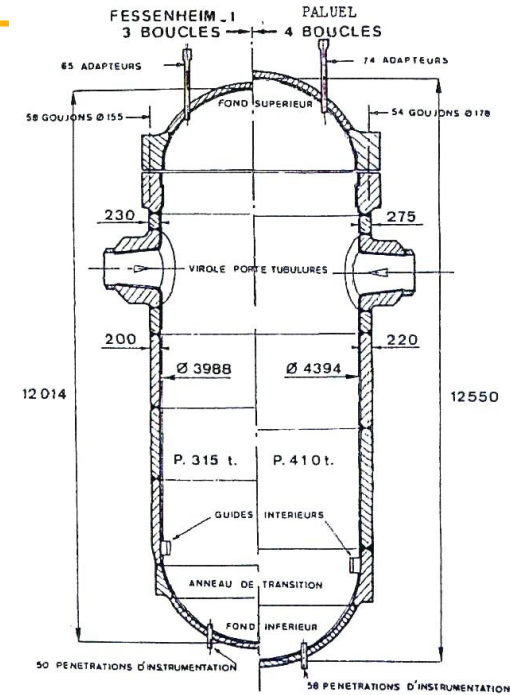
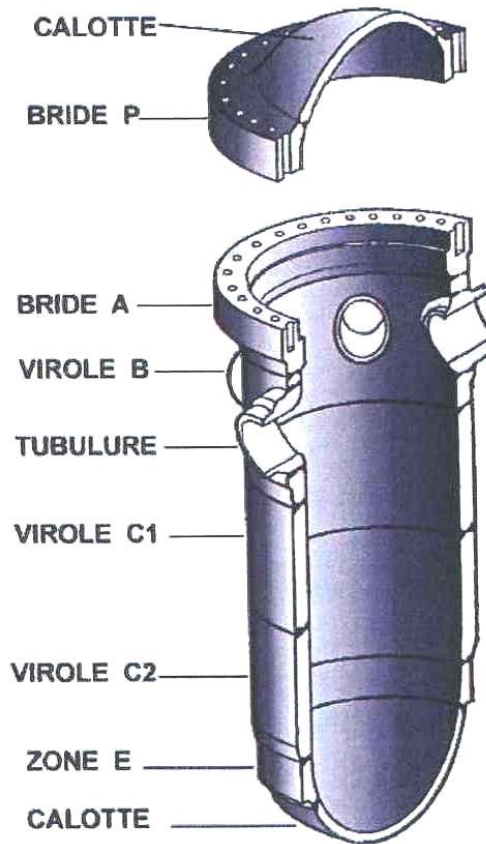
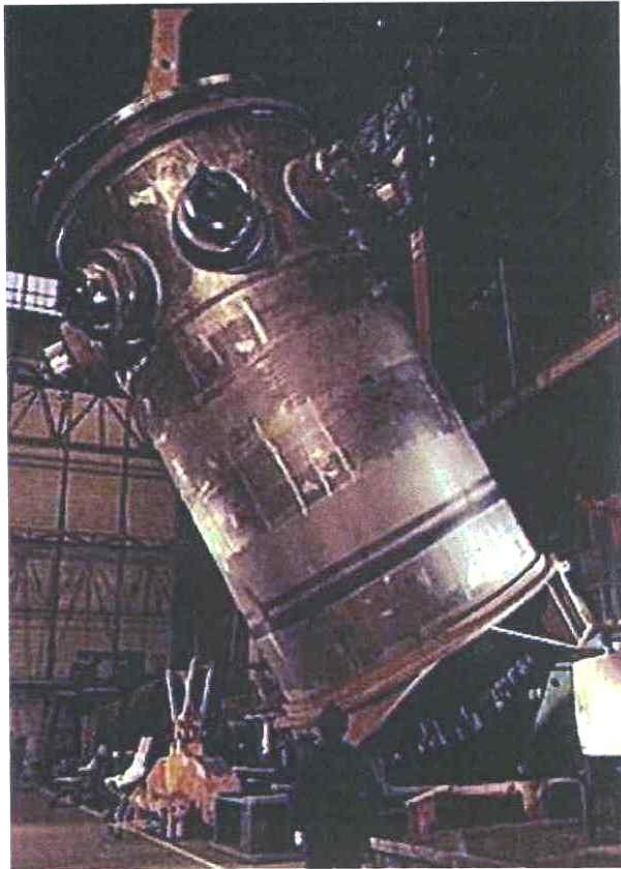


Long term evolution of point defects

- Point defects : vacancies and interstitials
- Under irradiation : creation of interstitial and vacancies (high V concentration)
- High mobility of interstitial low mobility of vacancy
- Anisotropic diffusion of the interstitials
- Vacancy and interstitial recombination
- Point defects clustering : loops
- Point defects elimination at sinks (dislocation, grain boundaries + loops)
- Complex evolution, depends on all the sinks present in the material, on temperature, on stress



The reactor pressure vessel

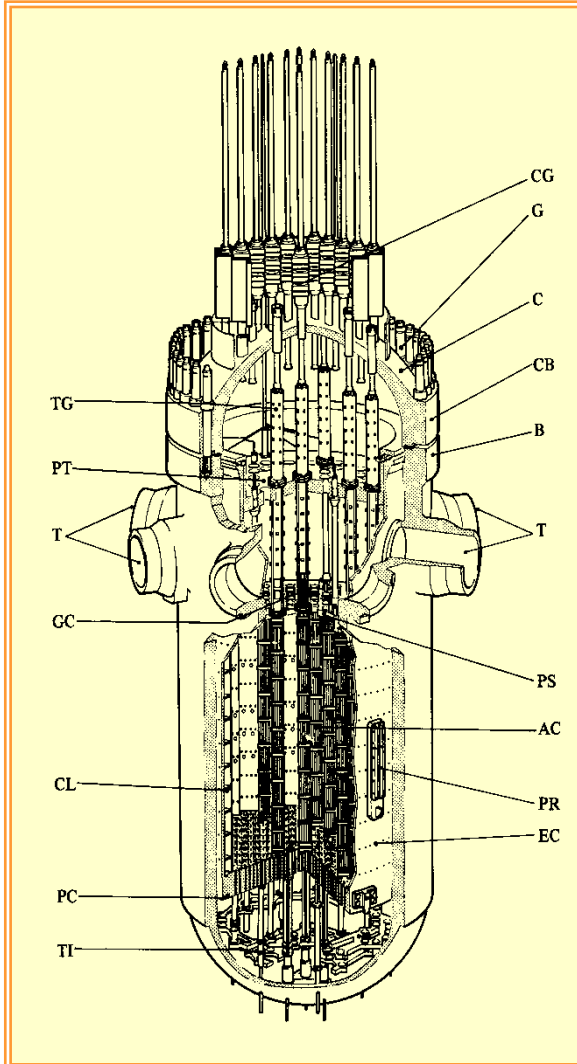


Functions of the reactor pressure vessel

The reactor pressure vessel is the second safety barrier

The pressure vessel is the only component which cannot be replaced

it has to keep its functions for the lifetime of the plant (environment, irradiation damage) in operating conditions, but also in accidental conditions...



Temperatures : 296 – 320°C

Coolant pressure: 155 bar

$\varnothing = 4400 \text{ mm}$

$e = 220 \text{ mm}$

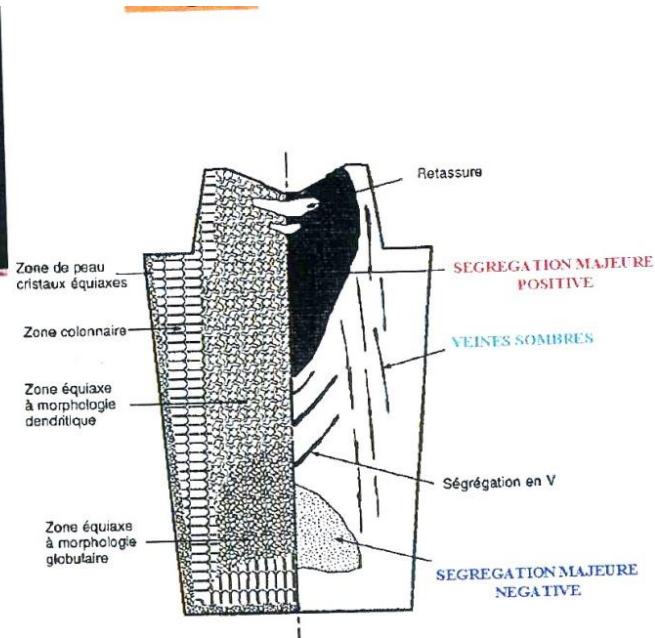
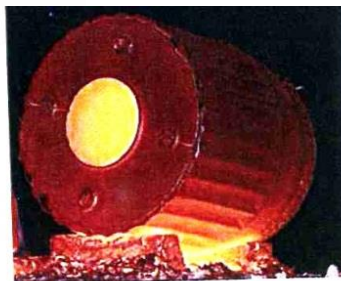
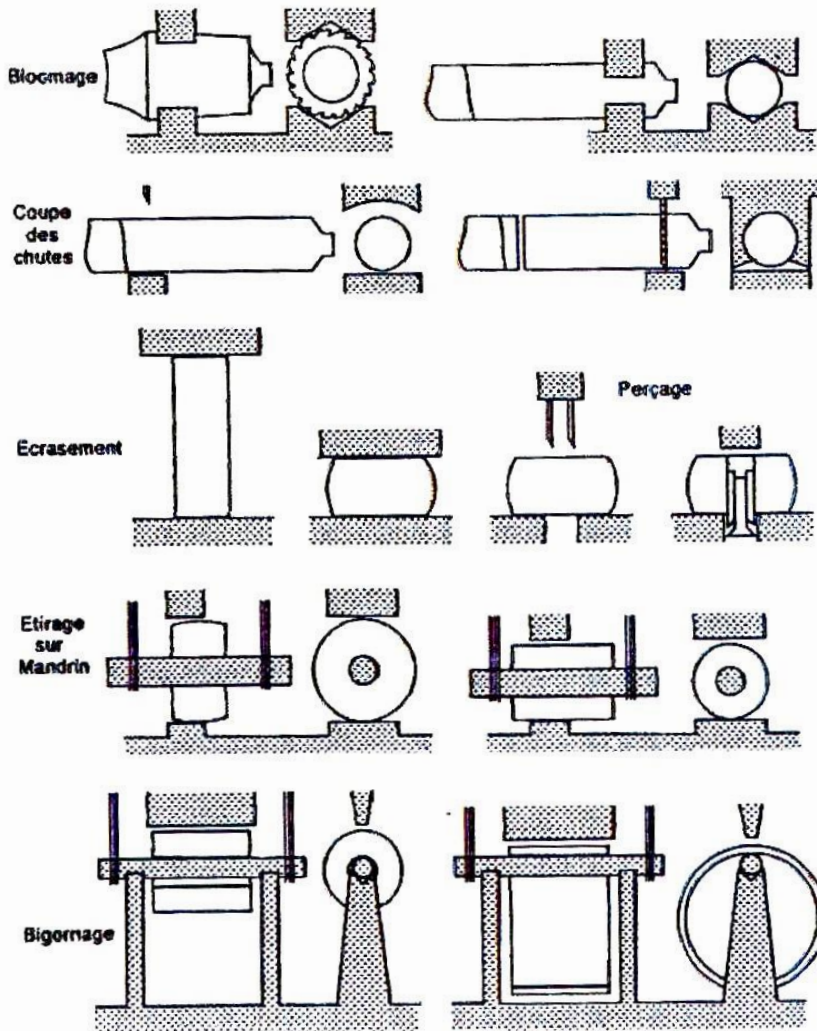
Gross weight 330 (900 MW) 440 t (1300 MW) EPR (520 t)

Steel A508 Cl. 3 = 16MND5
(bainitic steel)

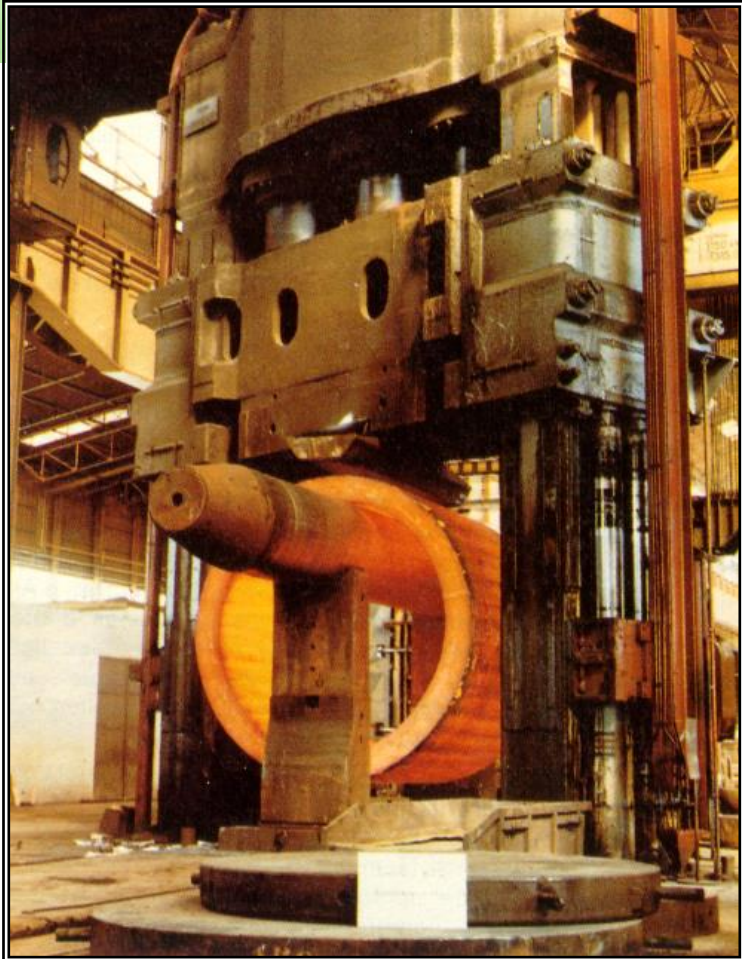
Internal cladding: 304 L = Z2CN1810
(austenitic stainless steel)



Fabrication process

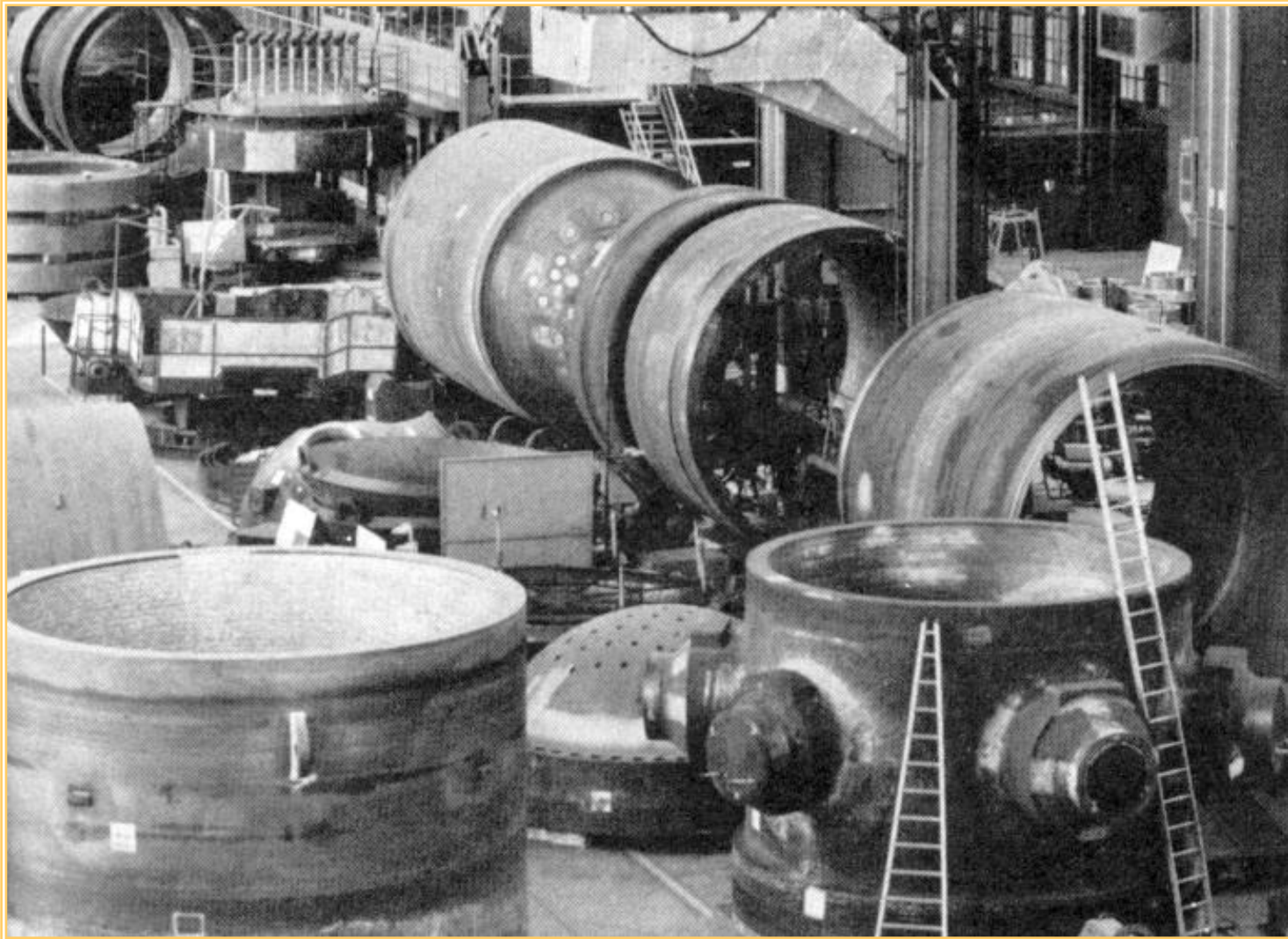


Forming of ingots



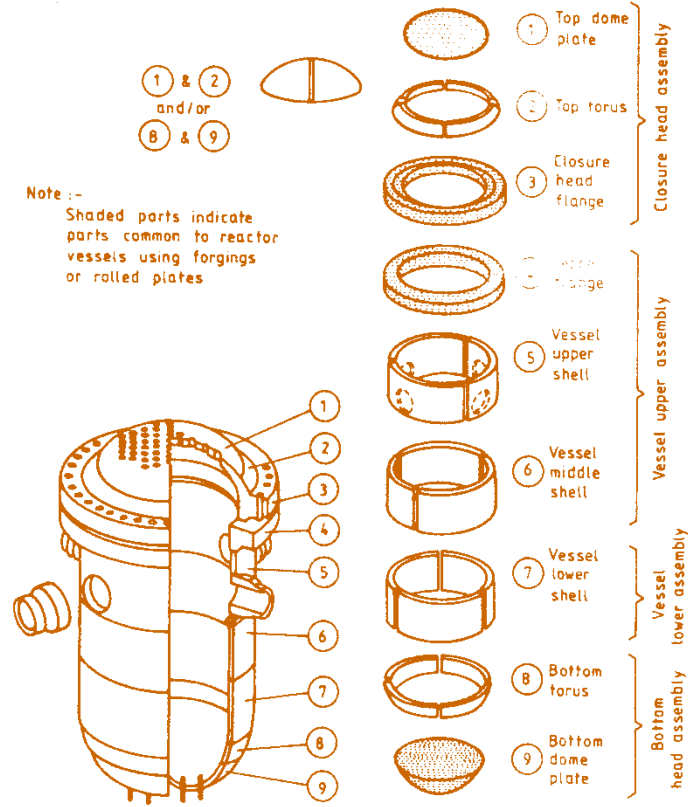
- Forging at 1100 - 1200°C
- Deformation ratio > 3
- Intermediate re-heating
- End of forming: hold at 600-650°C (H diffusion)
- Final heat treatments
 - γ -quench 850-920°C
 - tempered 635-665°C

Forged sections are welded



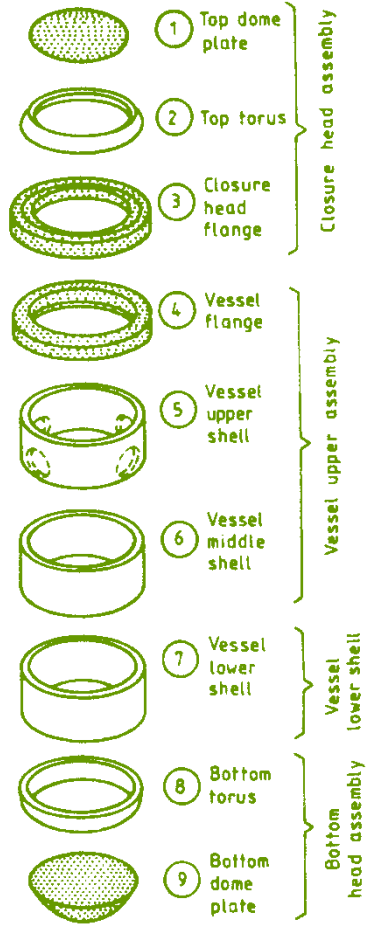
Assembly of the vessel

Rolled and welded flats



Westinghouse

Forged cylindrical rings



Le Creusot

PWR pressure vessel steels

Ferritic steel (bcc)
 16 MnNiMo5 (A 508 Cl 3)

Low carbon (easy welding)
 Strength by other elements, Mn....and structure (bainitic)

C _{max}	Mn	Ni	Mo	Cr _{max}	Si
0.2	1.15/1.55	0.5/.8	0.45/.55	0.25	0.1/.3

	P _{max}	S _{max}	Cu _{max}	Co
ppm	80	80	800	300

Low S, P et Cu to avoid irradiation induced embrittlement
 Low Co to avoid γ irradiation



Internal cladding for vessel

- To protect against the corrosion of the primary coolant: welding of a stainless steel layer (*beurrage in french*)
- Two layers are welded from planar sheets
 - 24 Cr, 12 Ni (Cr rich to compensate the loss due to dilution with base metal melted)
 - 20 Cr, 10 Ni (other layer similar to 304L-316L)
- Thickness: 8 mm (vessel steel: 220 mm)
- Formation of cracks under the layer
 - Cracking at low temperature (during the cooling, contamination due to H diffusion during the welding)
 - Pre- and post-heating

Introduction to fracture mechanics



Charpy

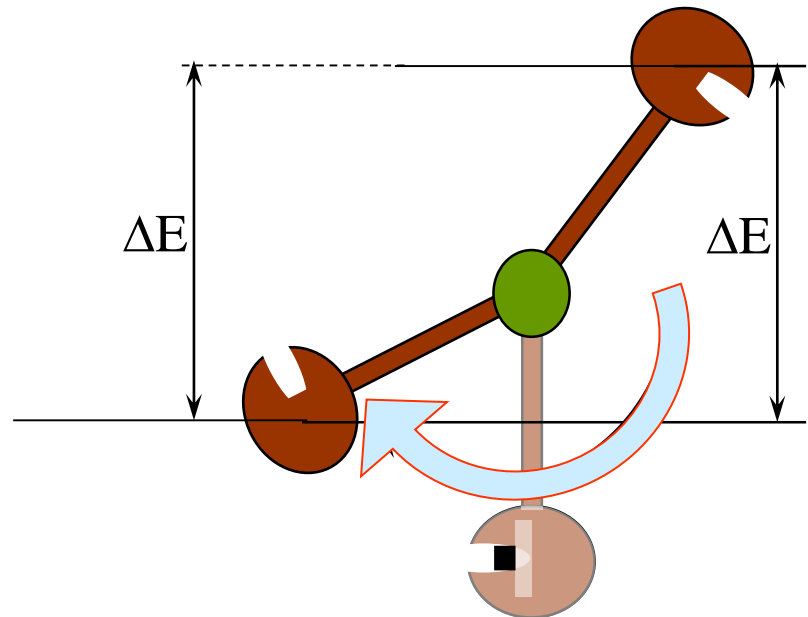
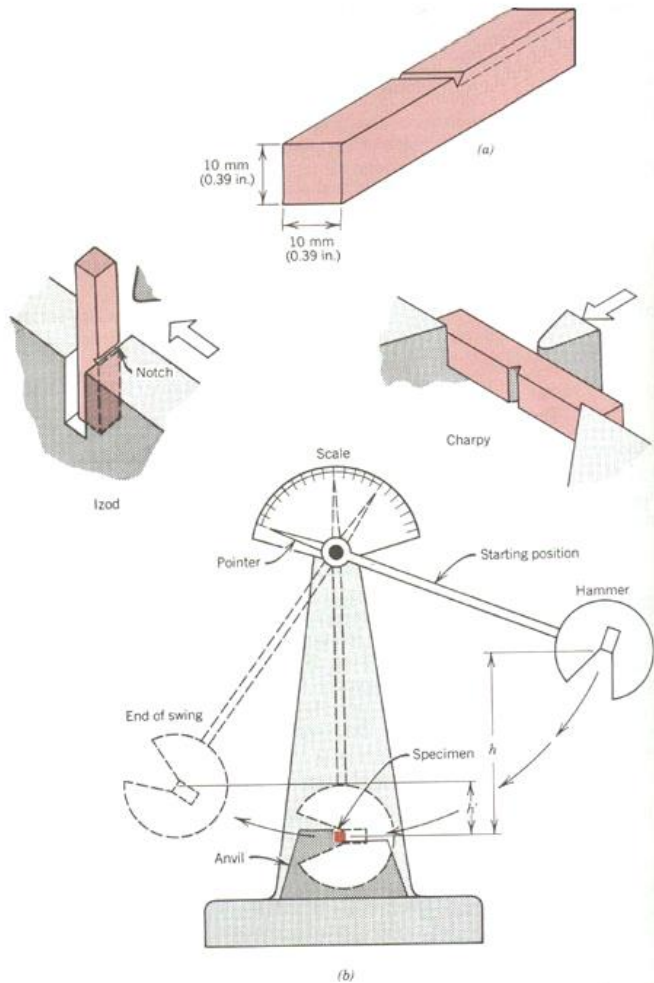
- Indication of a tendency to brittle fracture (resistance to cracking)
- Small samples
- Easy irradiation
- **Fracture toughness** (K_{IC})
 - Mechanical value of a resistance to crack propagation (design)
 - Much larger testing samples

Brittle - ductile transition in ferritic steels (bcc)



Quantification of the resistance to cracking

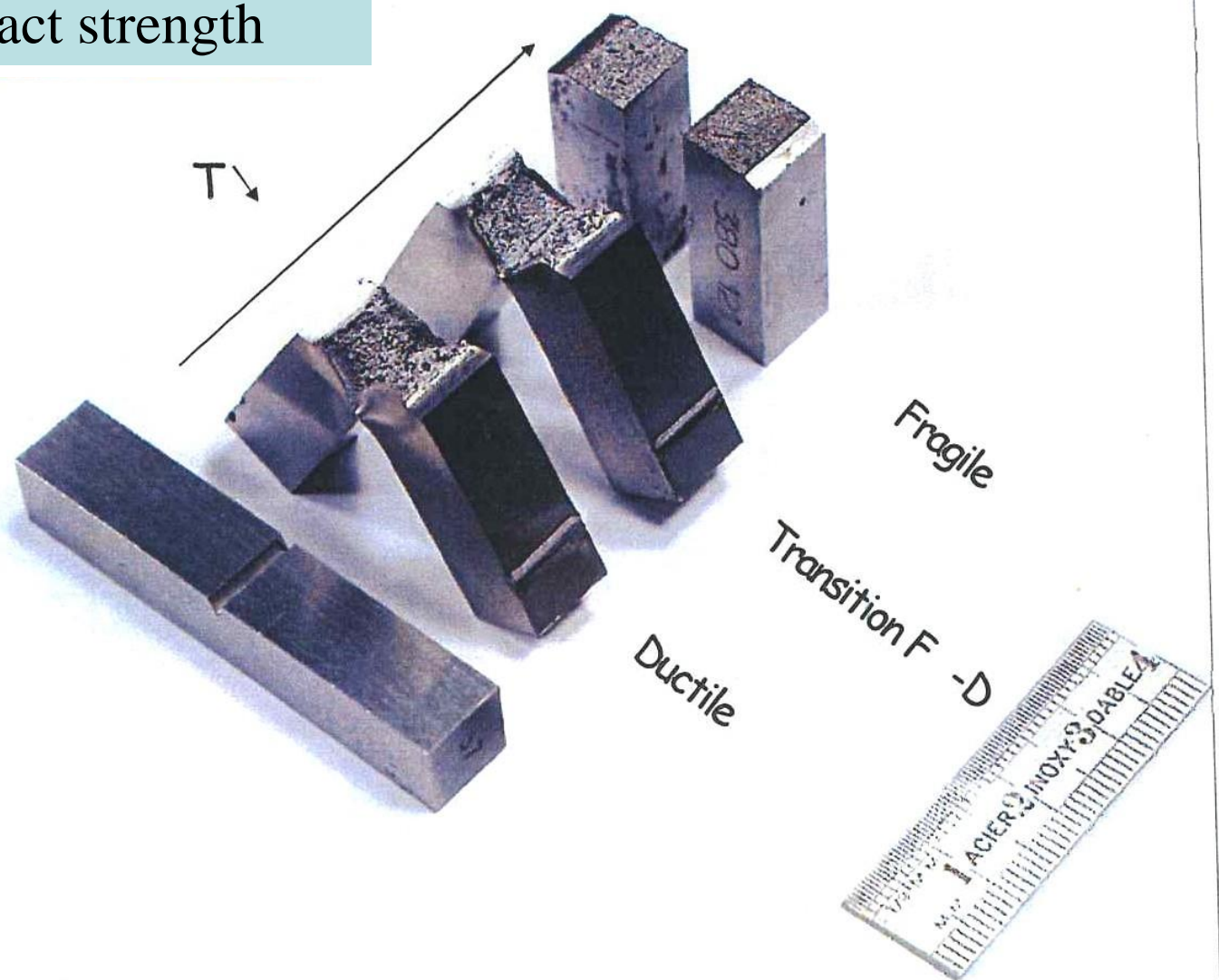
Measurement of the energy absorbed during the test



Charpy samples

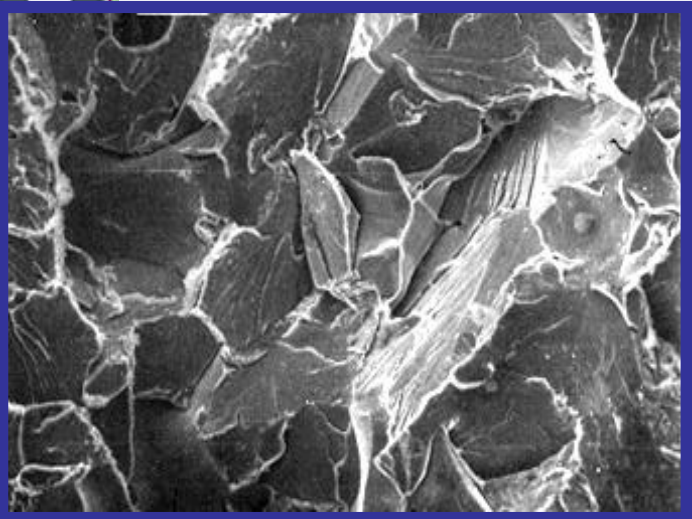


Impact strength



CHARPY
Test

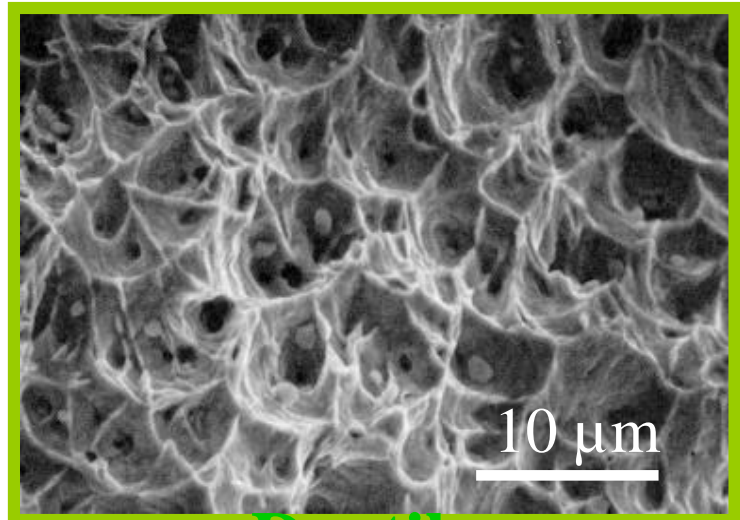
Fracture surfaces after Charpy impact test



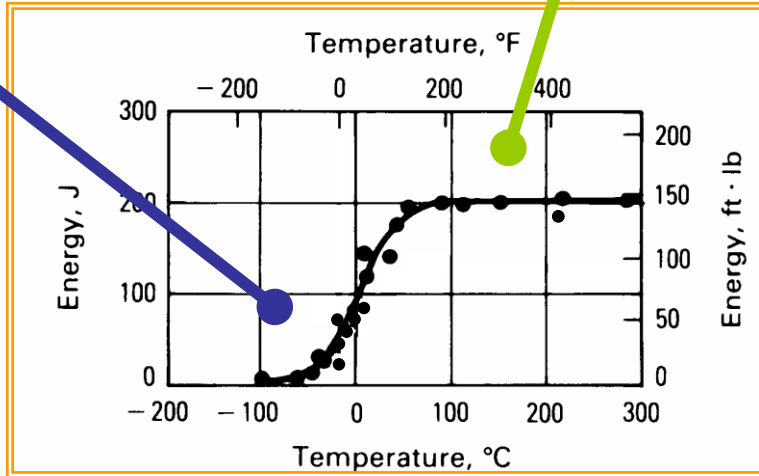
Brittle

Ductile : plastic deformation and rupture after coalescence of cupules, high energy absorbed during the tests

Brittle : cleavage = de-cohesion of the crystal along specific crystallographic planes, low energy absorbed during the test before rupture



Ductile



Evolution of the absorbed energy during the test as a function of the temperature

Fracture toughness (K_{IC})

Resistance to crack propagation

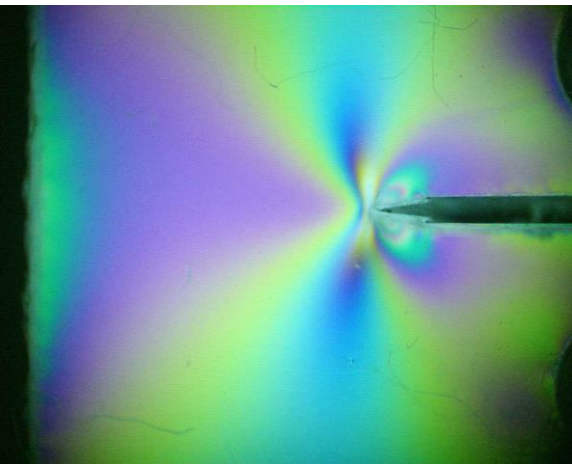
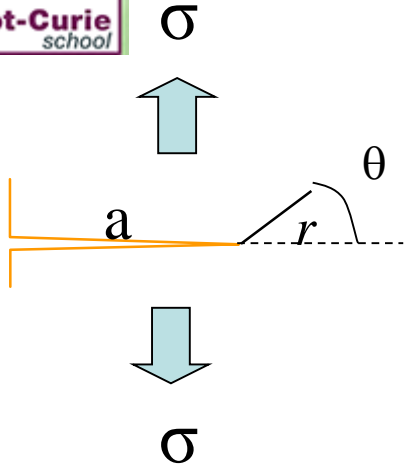
At the root of a crack, the stress diverge
=> equal to infinity at the tip

$$\sigma_{loc} = \frac{K_I}{\sqrt{2\pi \cdot r}} \cdot f(\theta) \quad K_I = \sigma \cdot \sqrt{\pi \cdot a}$$

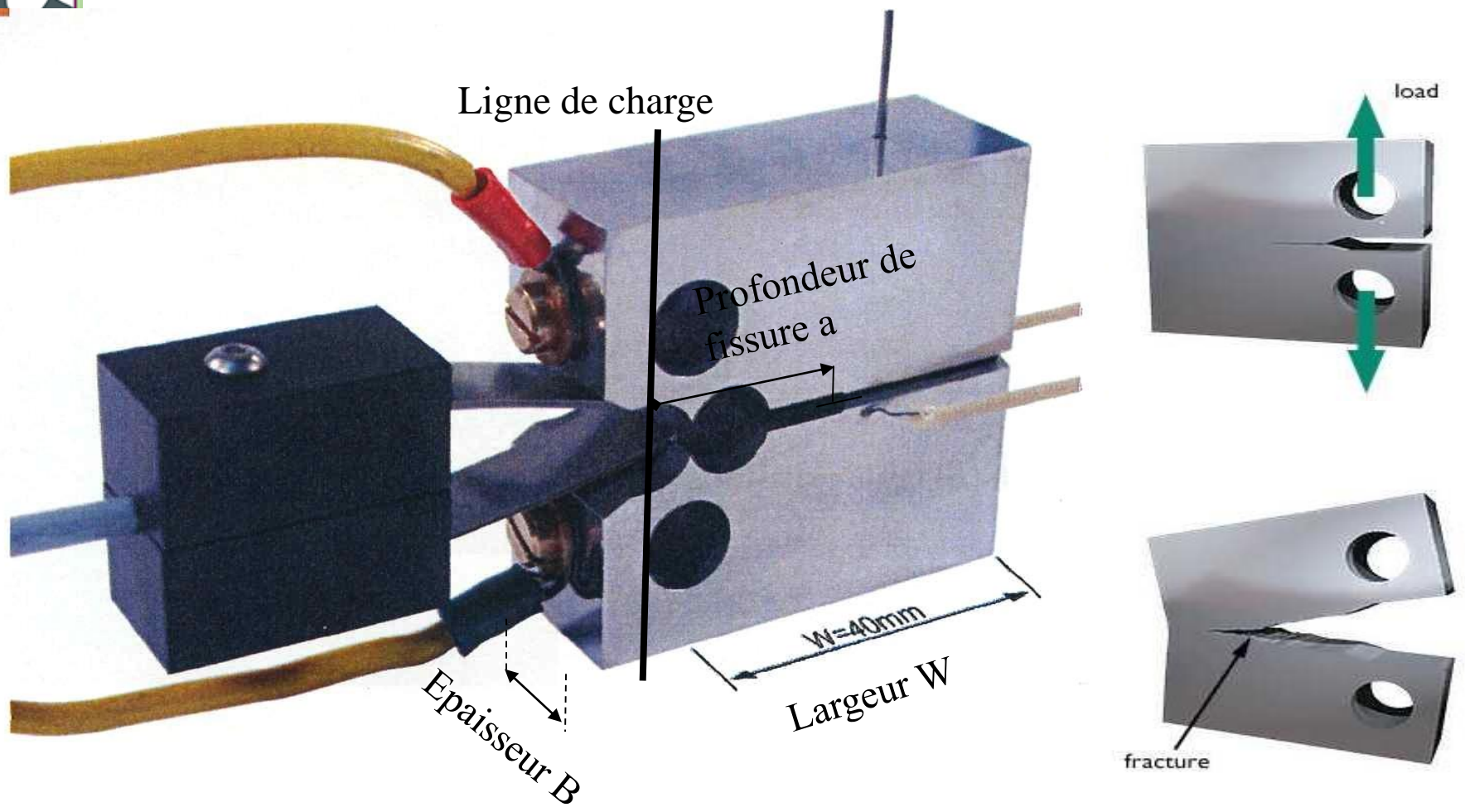
The stress intensity factor K_I allows to give a value to the singularity (in $\text{MPa} \cdot \text{m}^{1/2}$)

The resistance of a material to crack propagation is measured by a toughness test. Fracture at $K_I = K_{IC}$ (ASTM E 399) (no propagation if $K_I < K_{IC}$, *if $K_I > K_{IC}$ than the crack propagates suddenly*)

Knowing the crack geometry and the stress state, allows to forecast crack propagation or not.



Fracture toughness: resistance to crack propagation

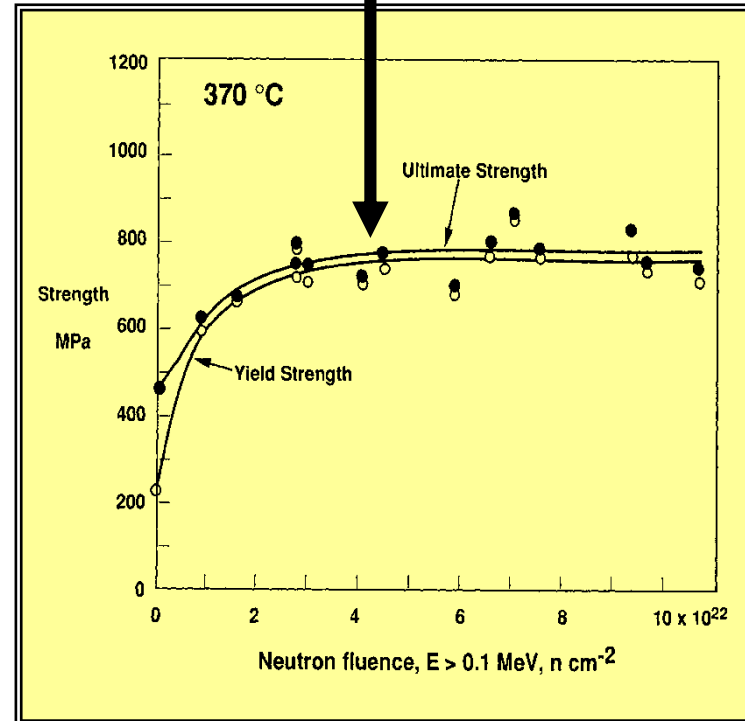


Increase in yield strength due to irradiation



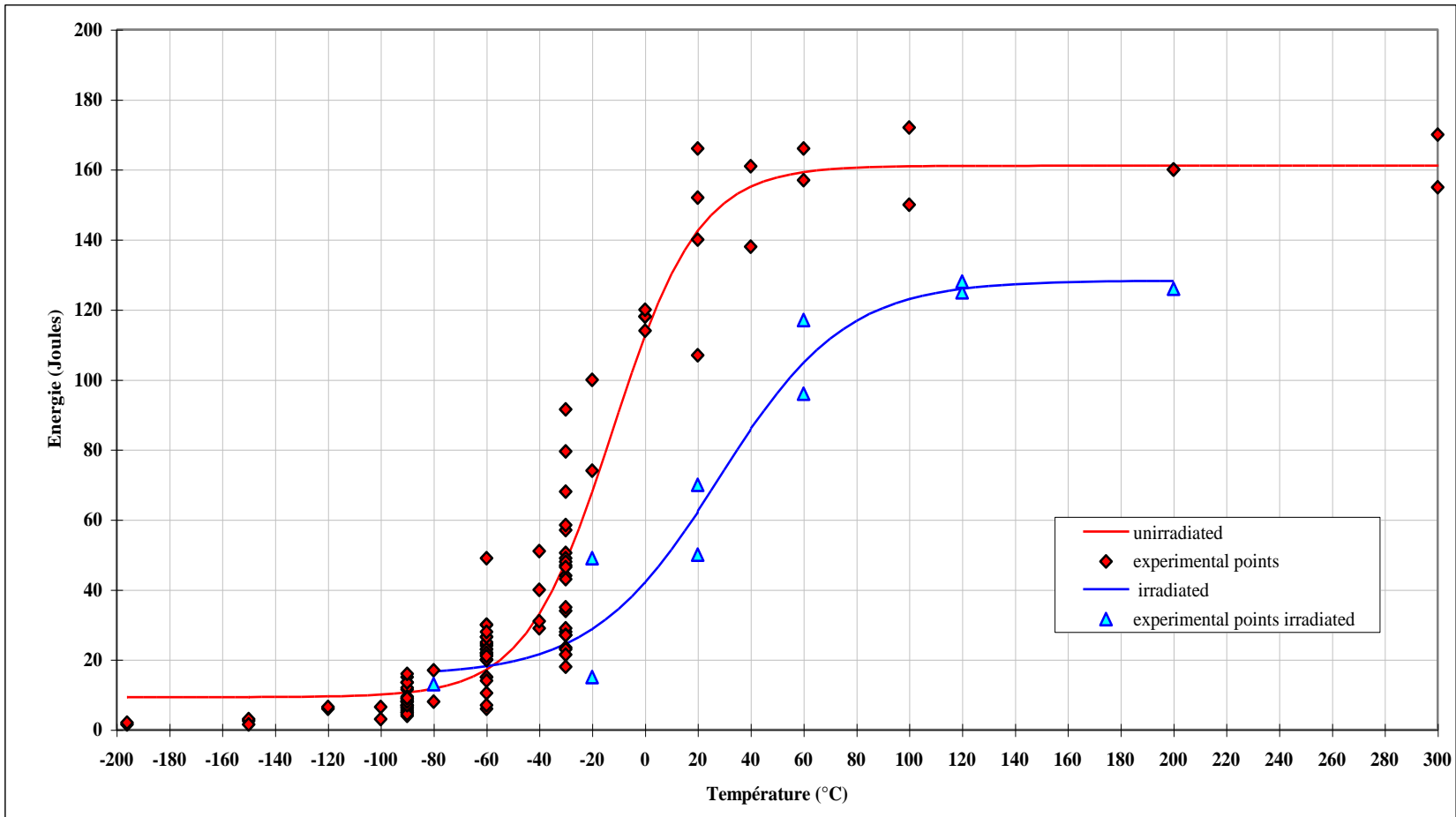
- Pressure vessel irradiation: 10^{-10} dpa.s⁻¹
- Dislocation loop formation, and other defects like clusters of minor elements induce an increase in mechanical properties.

Saturation of the hardening due to the saturation of the microstructure (saturation of defects density)



Evolution of ductile to brittle transition temperature

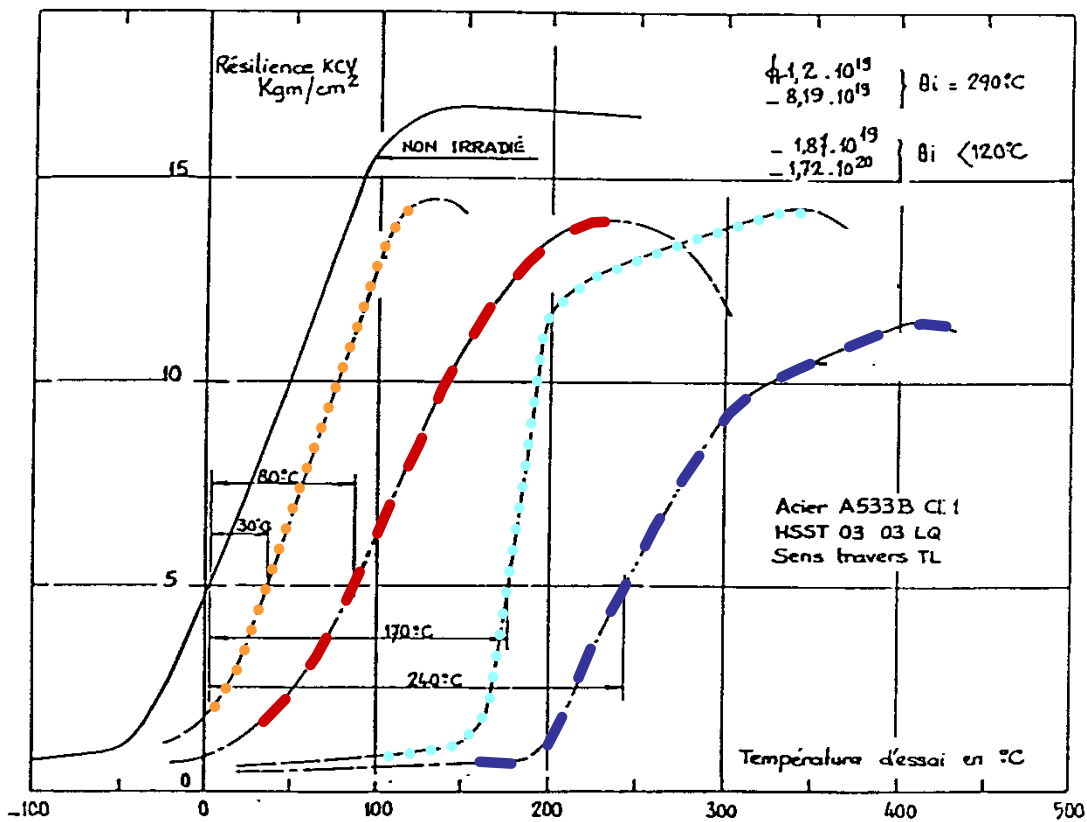
Higher strength increases the probability of failure by cleavage, leading to a higher transition temperature.



Impact of irradiation temperature

At higher temperature, continuous recovery occurs during irradiation and embrittlement is reduced.

For the same irradiation dose, the effect is higher at 120°C than at 290°C



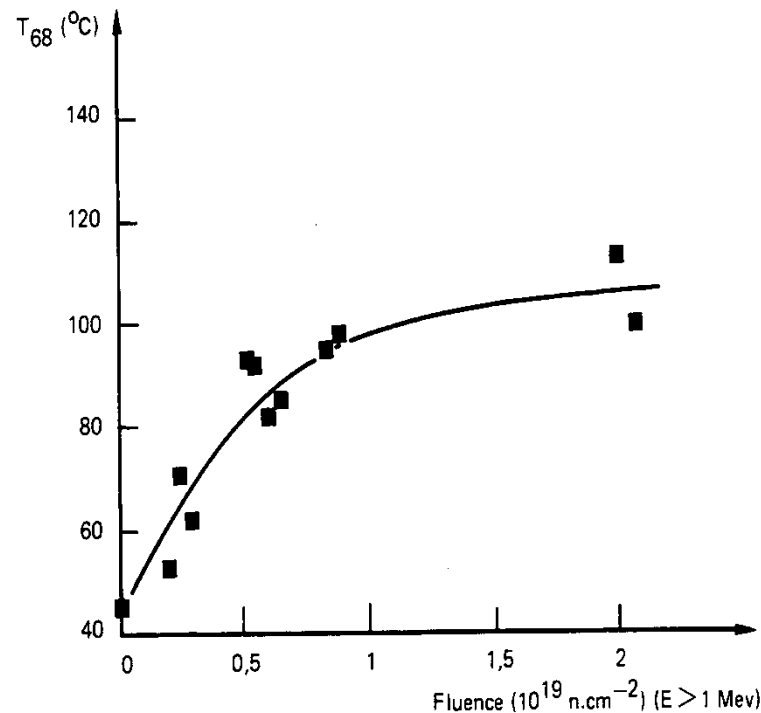
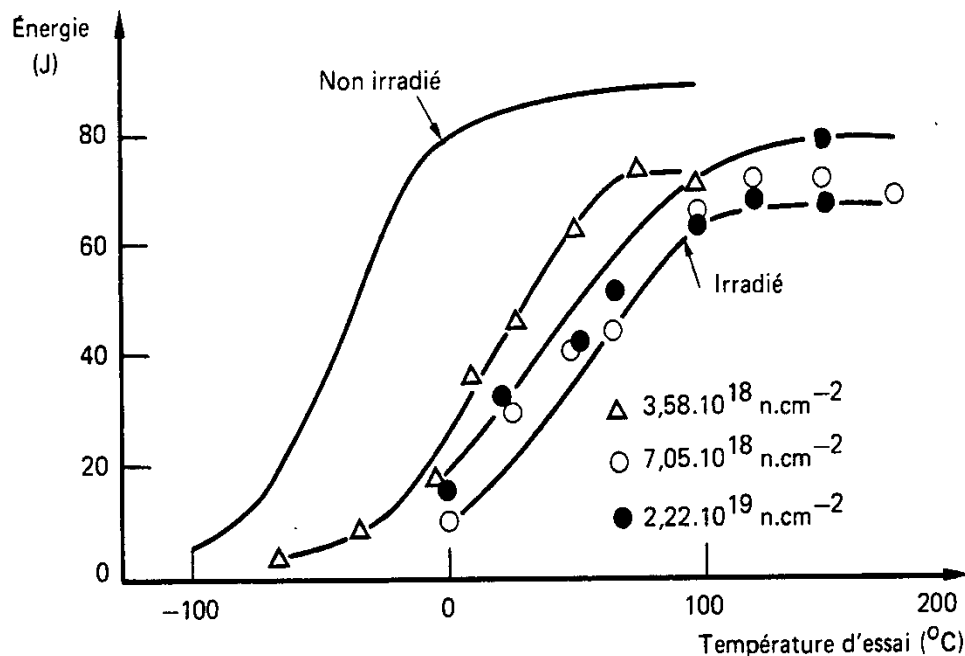
Tirr. = 290°C

Tirr. = 120°C : lower elimination of PD, less recombination, more hardening

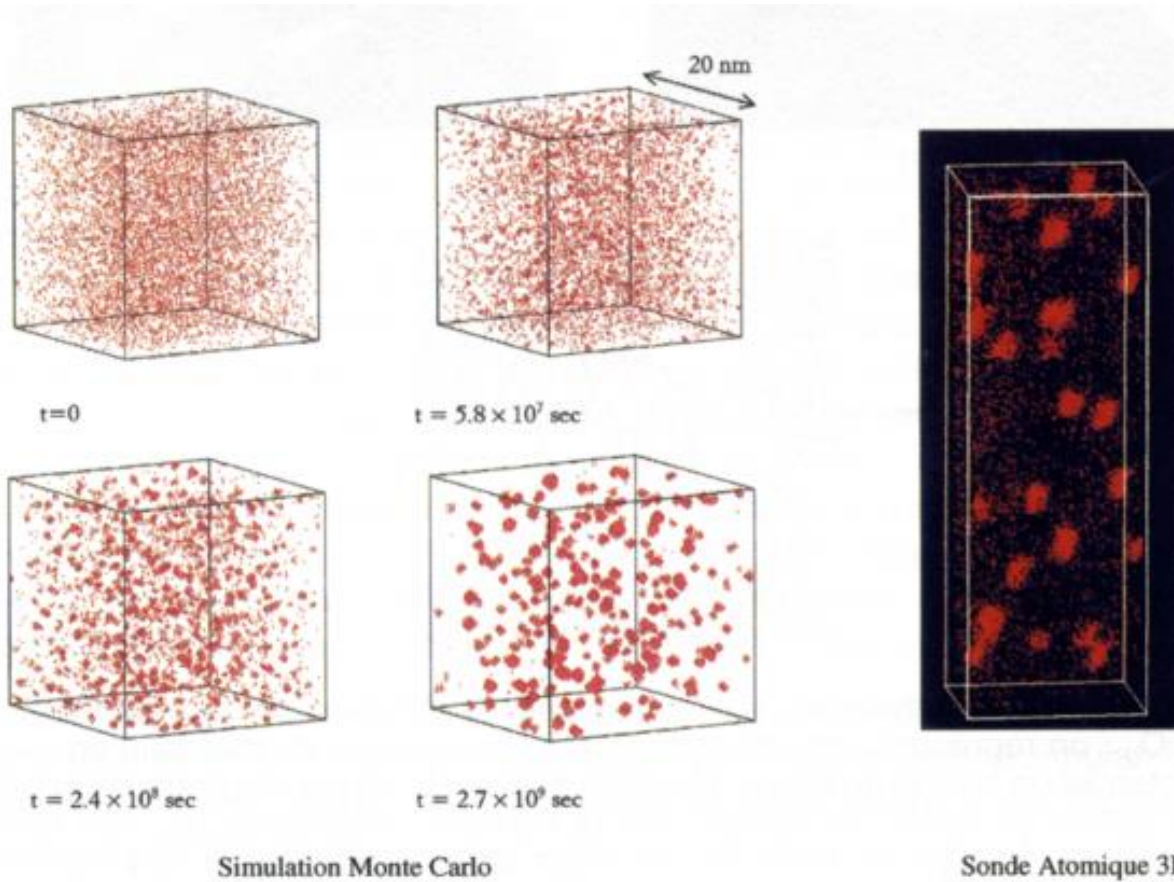
Saturation of irradiation effects

Ductile Brittle Temperature Transition increase vs fluence

Induced by hardening and also the formation of précipitates



Non linear increase of the **DBTT** as a function of the fluence, rapid saturation



On each crystal site
 Fe or Cu
 Thermodynamics for
 interactions
 between species
 Kinetics, according to
 probability of
 occurrence
 Simulated time: One
 century

Behavior of real pressure vessel steels

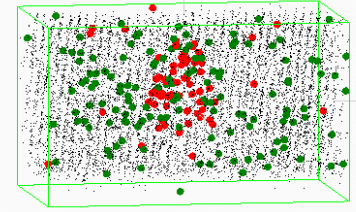
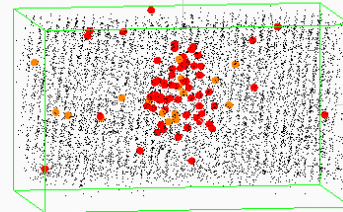
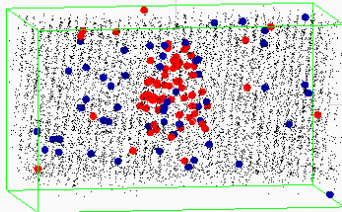
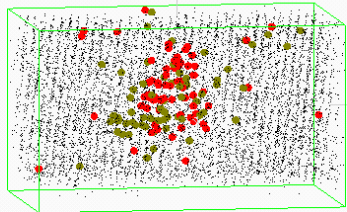
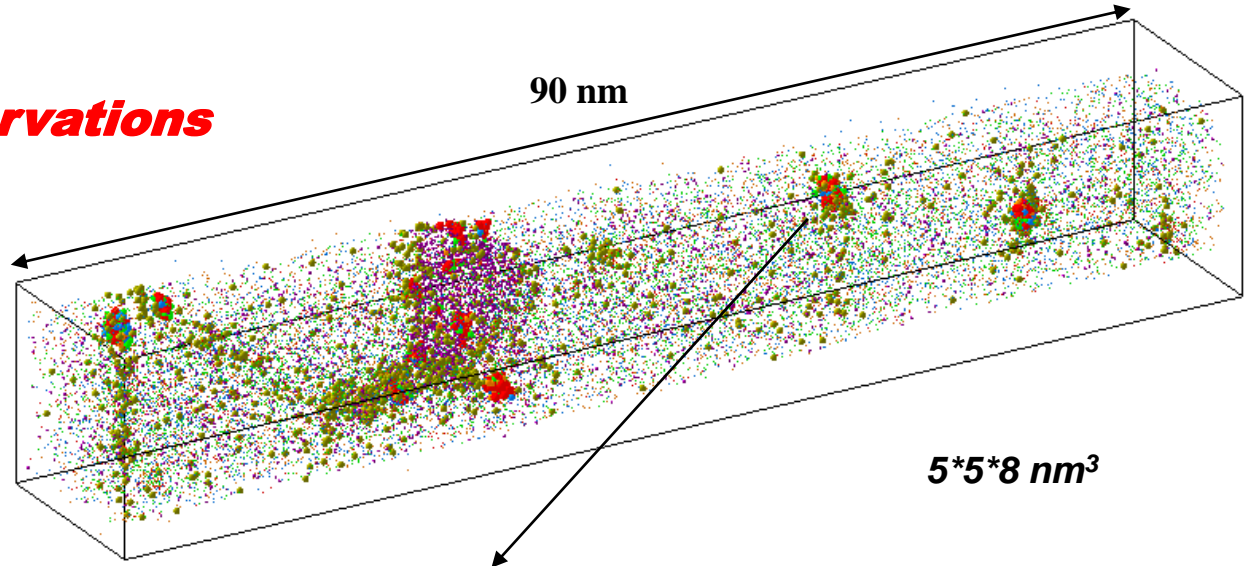


More complex chemistry

3D-Atom Probe observations

flux : $1,5 \cdot 10^{15} \text{ n.m}^{-2} \cdot \text{s}^{-1}$
fluence: $9,7 \cdot 10^{23} \text{ n.m}^{-2}$
 $T \approx 300 \text{ }^\circ\text{C}$

Diffuse solute clusters
 $2 \text{ nm} - 5 \times 10^{23} \text{ m}^{-3}$



Fe

bal

Cu

20

P

3.5

Mn

4

Ni

3.5

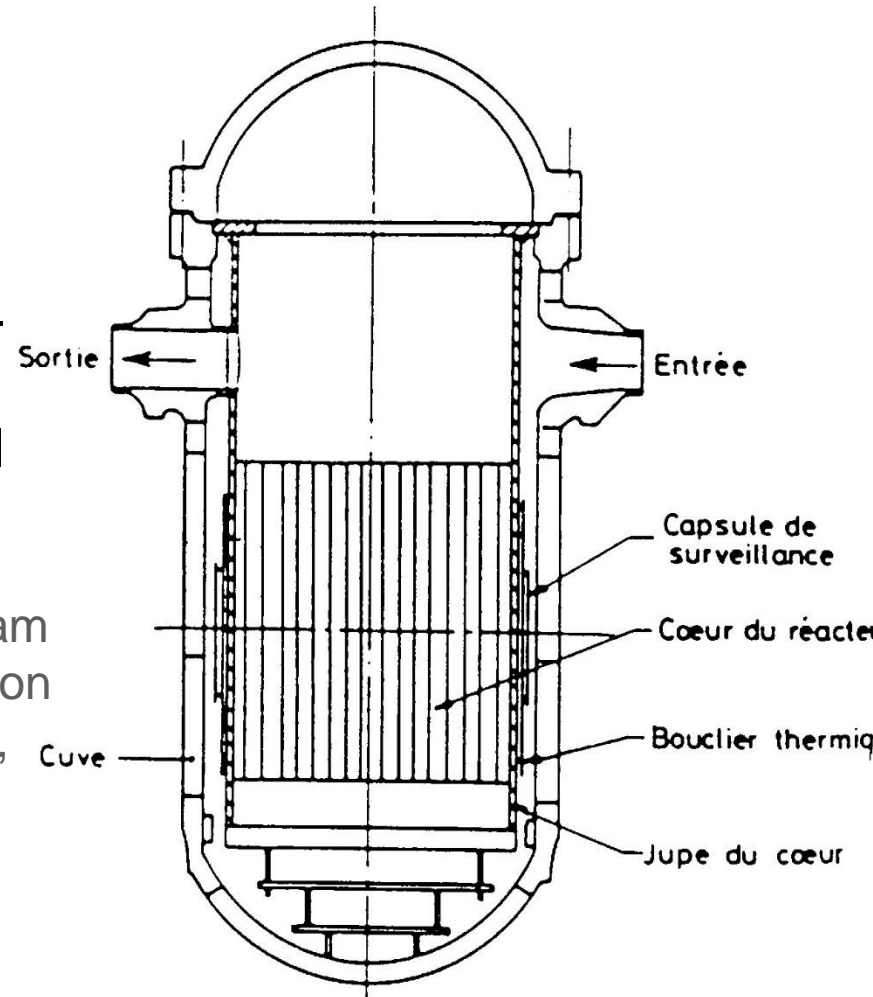
Si

3.5

Surveillance Program

Tensile and Charpy-V specimens are located at the periphery of the internal structures.

The materials submitted to the surveillance program are the central part of the vessel, the C1-C2 weldment with its HAZ and a reference material common to all French nuclear power plants.



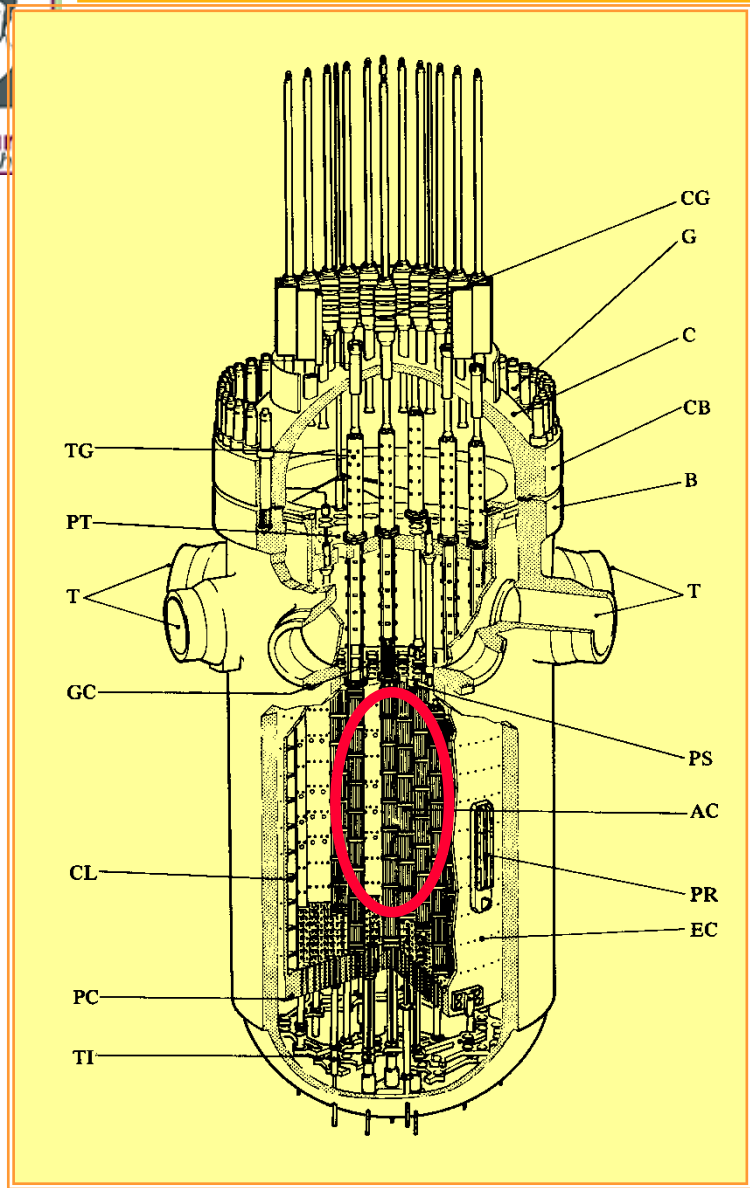
Regulations stipulate that the surveillance program must be representative regarding the irradiation conditions (temperature, neutronic spectrum, ...) and the materials

Base metal taken from overlength of the shell

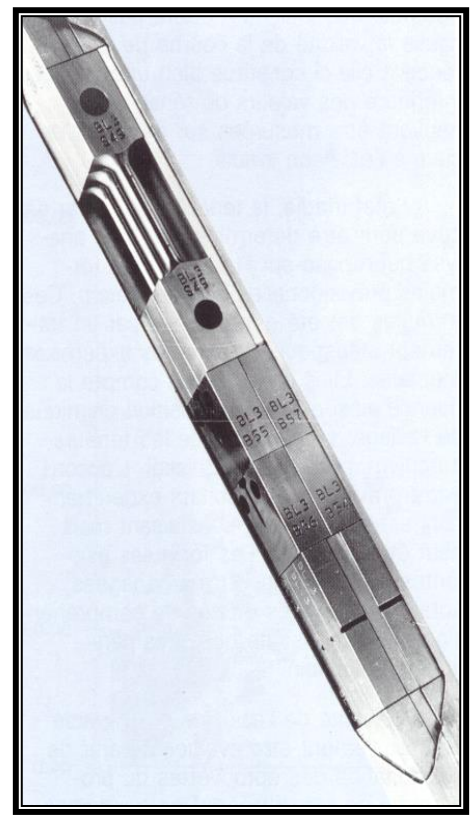
Weld and heat affected zone elaborated under identical welding conditions as the core zone components



Irradiation devices



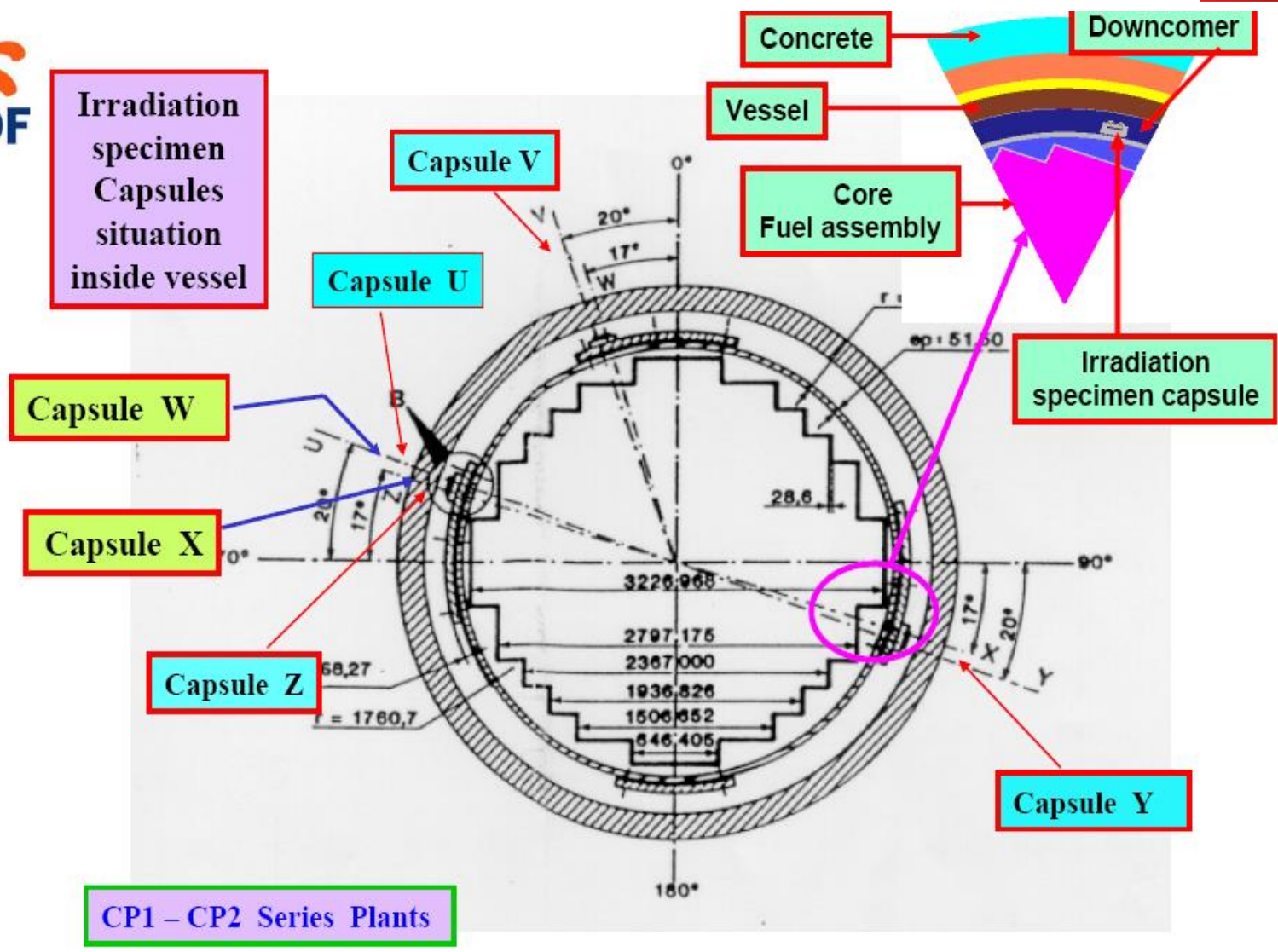
4 holders
 Removed every 1/4
 quarter design life
 4, 7, 10 and 14 years
 Fast neutron dose
 integrators
 Samples: Charpy,
 tensile, K_{IC}



Surveillance programme



Irradiation specimen Capsules situation inside vessel



CP1 - CP2 Series Plants

	Capsule W (u)	Capsule (v)	Capsule X (z)	Capsule (y)
<i>Time of duration in vessel (years)</i>	4	7	9	14
<i>Equivalent time of irradiation of the Vessel (Years)</i>	11.2	19.5	28.0	39.1

Extension of the RPV irradiation surveillance programme based on the introduction of reserve irradiation capsules is engaged on the French plants since 1999 for all reactors.

Two reserve capsules W and X in place of capsules U and Z after removing from reactor these capsules U and Z.

Materials and fluxes



- Largely inspired from American regulations
- Fulfills the french safety Authorities requirements
- For all 900, 1300 and 1450 MWe reactors, the core zone is generally made up of two shells and the associated weld. As these materials experience a marked embrittlement under their design end of life (40 years), mechanical properties and particularly the rupture characteristics through impact strength are monitored for each reactor zone
- Materials positioned in capsules at locations well characterized for temperature and neutronic conditions
- **Neutronic flux is higher (x3) than the one undergone on the vessel in order to anticipate the embrittlement.** The flux is evaluated for each capsule through a lead factor, corresponding to the ratio of the neutron fluxes of more than 1 MeV energy undergone by the capsule and the vessel at the most irradiated point.

Material Ageing under Irradiation

Embrittlement Formulas :

The general form is indexed on the temperature shift of Charpy-V :

$$\Delta RT_{NDT} (\text{°C}) = CF \cdot \phi^G$$

where CF is the Chemical Factor, ϕ is the fluence (10^{19} n/cm² with $E > 1$ MeV) and G an exponent.

Formulation based on steels and welds from the French nuclear program :

This particular formulation is for the highest effect of irradiation (FIS).

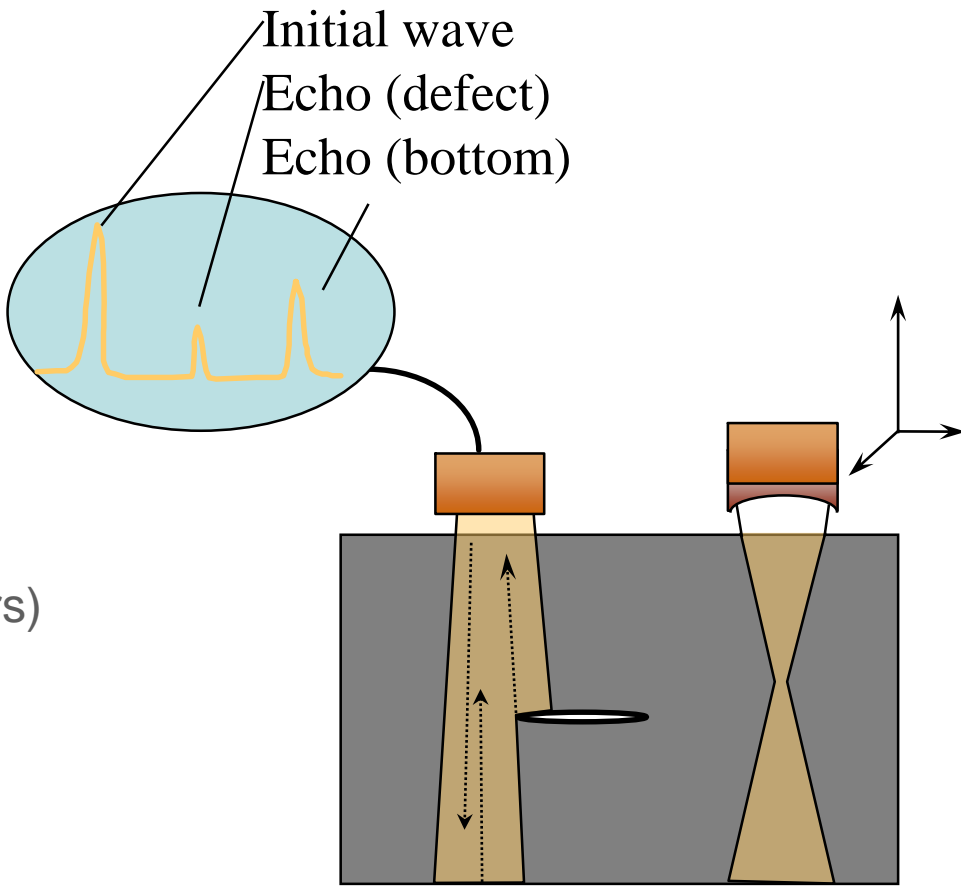
Another formula is available in RCC-M for mean irradiation effect (FIM).

$$\Delta RT_{NDT} (\text{°C}) = 8 + \left[24 + 1537 \cdot (P - 0.008) + 238 \cdot (Cu - 0.08) + 191 \cdot Ni^2 \cdot Cu \right] \cdot \phi^{0.35}$$

Non destructive examination (NDE)

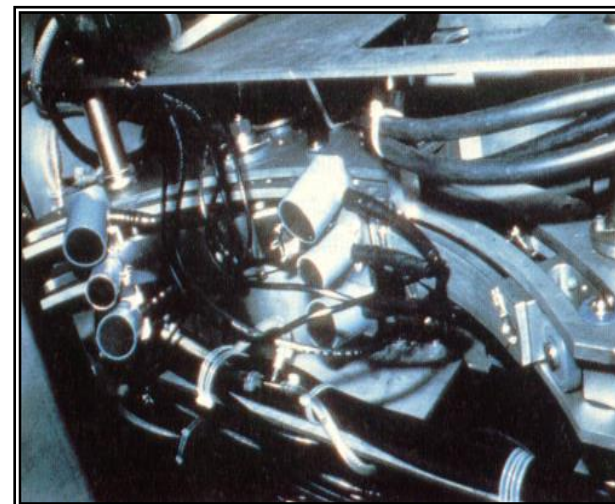
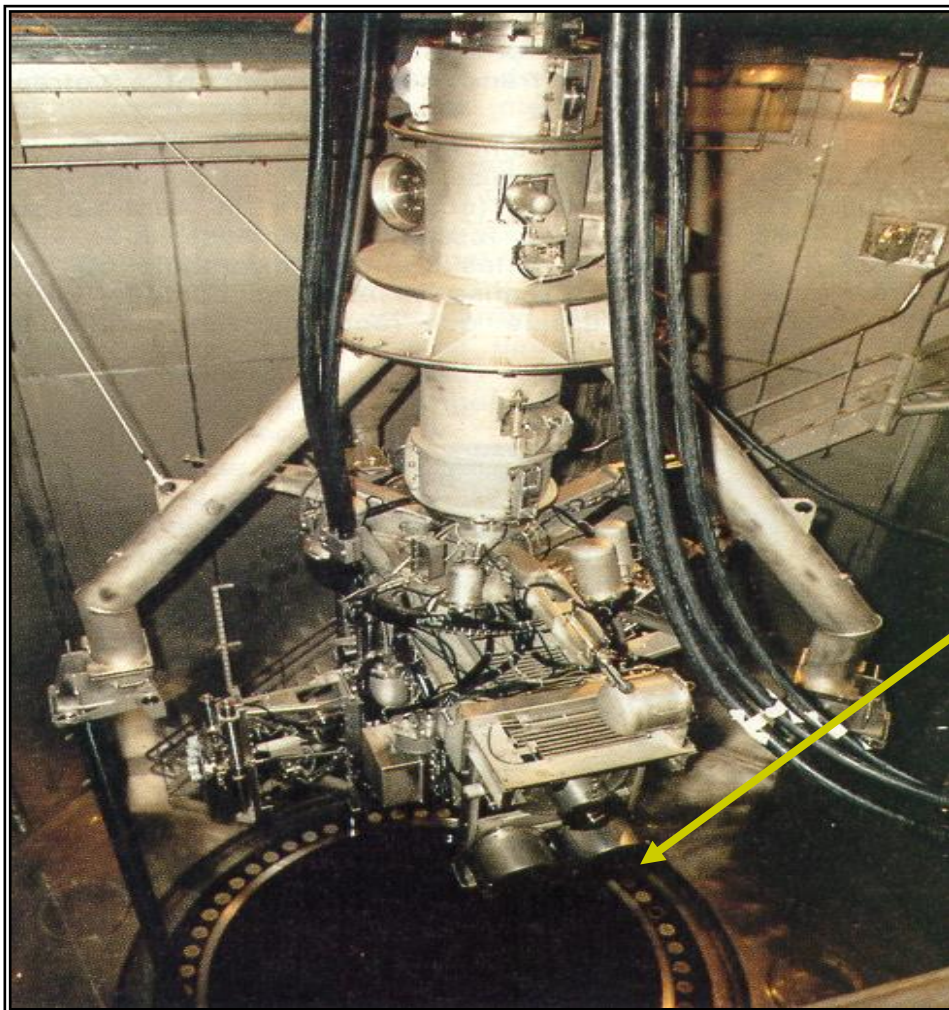
Ultra sonic testing

- Elastic strain wave
- Under water
- Focalized probes
 - x, y, z examination
 - 3D description of the cracks
 - A few mm accuracy
 - Inspection techniques (5 years)



In service NDE

US probes



Inspection device for a PWR

Accuracy < 1 mm

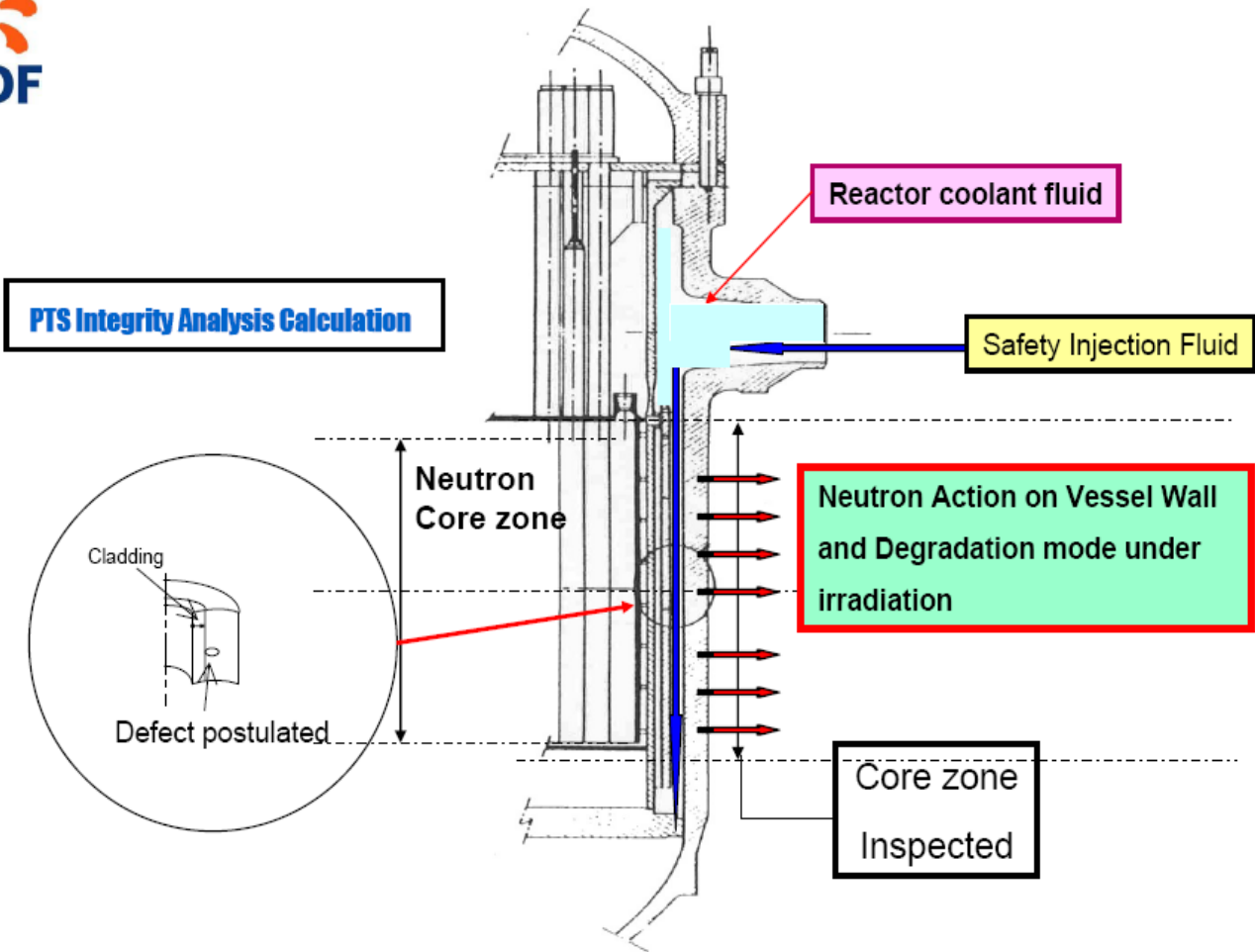
Defect sizes:

6 - 7 mm

CPO: 31 known defects

CPY: 2 known defects

Assessment of vessel integrity



Assessment of vessel integrity

Approach considered in relation with French codification

Demonstration of the integrity of the vessels in all conditions of loading, parameters :

- the RT_{NDT} ,
- all parameters fluence, defect distribution, transient, temperature.

The most severe conditions to be considered is the pressurized thermal shock (PTS) taking account for hypothesis shallow cracks beneath the cladding (subcladding area).

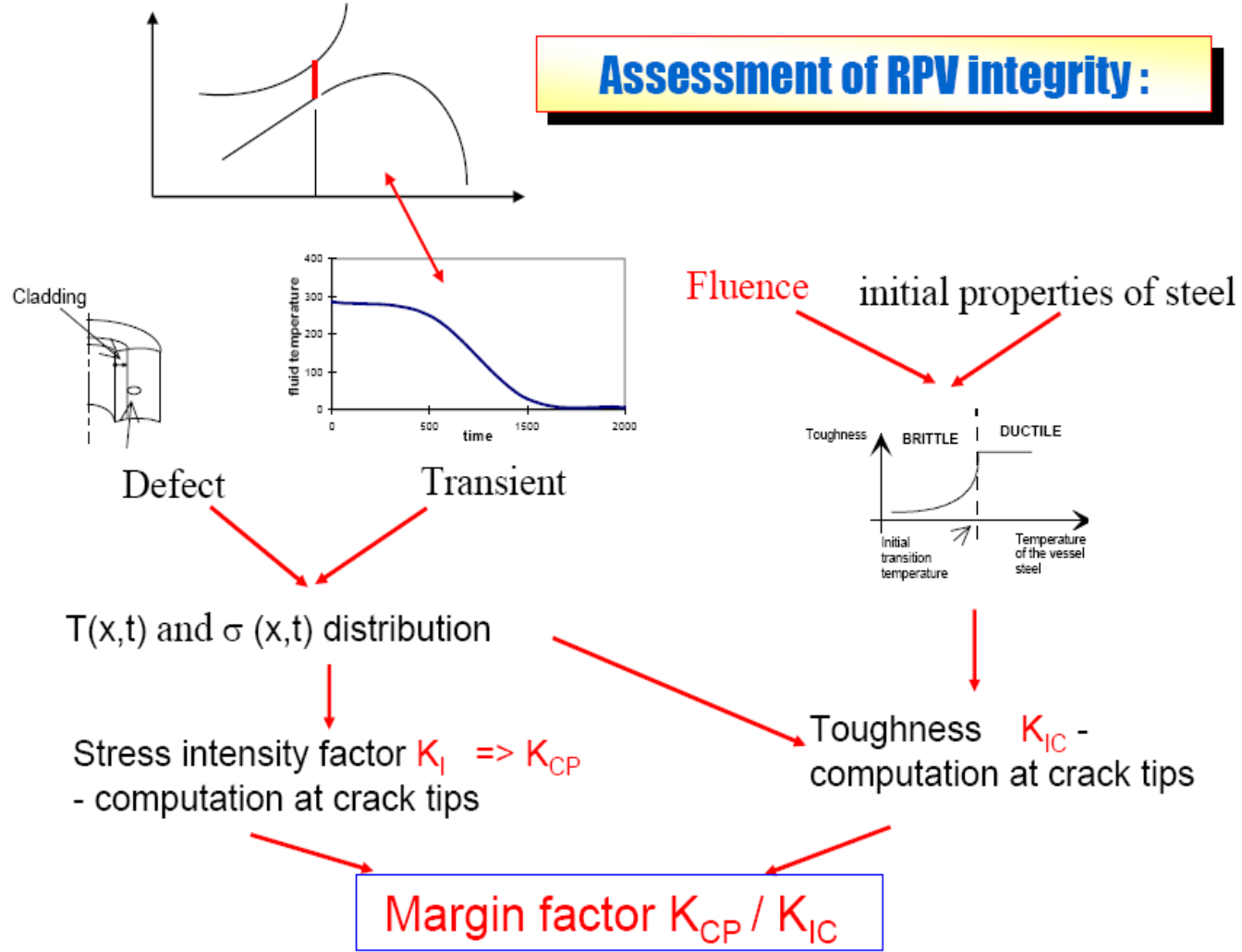
Justification of the Vessel integrity

Demonstration of the margin on

- brittle fracture
- ductile fracture (ductile tearing).

Assessment of vessel integrity

Assessment of RPV integrity :



Extended life of PWR vessels

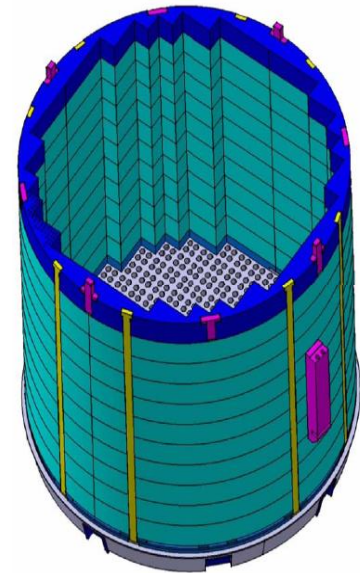


Old reactors

- Fuel management for reduced leakage
- Thermal recovery of pressure vessel
- Warm safety water tank

New reactors

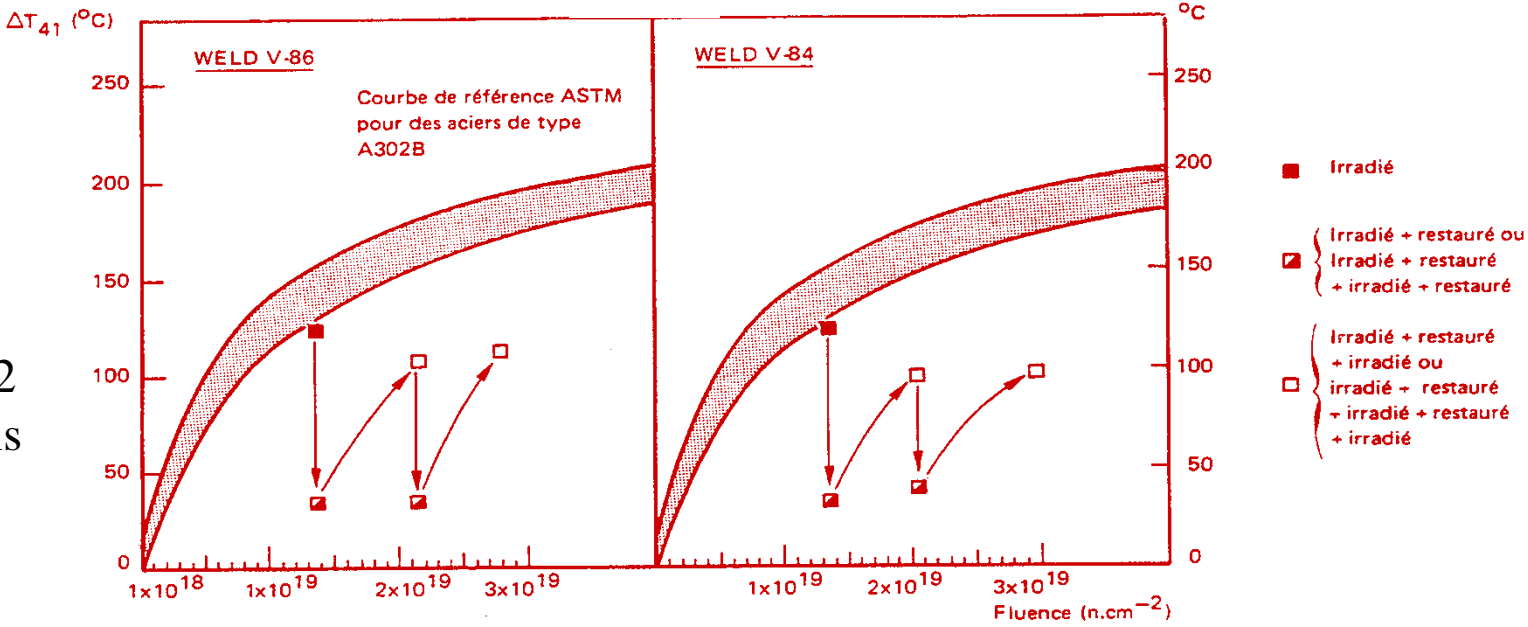
- Fine chemistry control during processing
- Reduced neutron flux by design (EPR heavy baffle)



Irradiation damage recovery

The clusters with different chemistry can be dissolved by intermediate temperature annealings. Yield strength increase is recovered as well as toughness.

This operation is performed in Russia where vessel steels are highly irradiated (2 x PWR, diameter is lower)



VVER: works at lower temperature ~ 270 °C and [Cu]<0.15% [P]<0.025% higher, RT_{NDT} increases faster than PWR for instance

Pressure vessel conclusions

Important metallurgical factors influence the properties, in particular the toughness, of French PWR steel (16MND5).

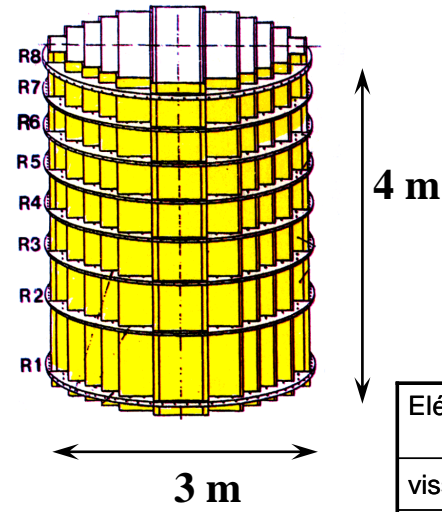
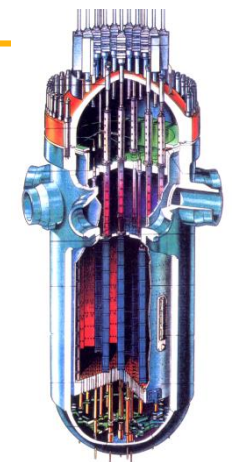
Careful control of elaboration, of microstructure and embrittlement under neutron irradiation are the key of good results.

The surveillance program warrants that no unpredicted deviation occurs.

Comprehensive work on the basic mechanisms at the origin of Copper embrittlement is also under investigation with cascade dynamic simulations and to be extended to get closer to real compositions

Enormous stakes to increase the lifetime of the vessel

Internals



Elément	Matériau	Dose (dpa) après 40 ans
visserie	Cold worked	0 à 80
cloisons	Quench annealed hypersimpe	10 à 80
renfort	Quench annealed hypersimpe	5 à 60

Pressure vessel

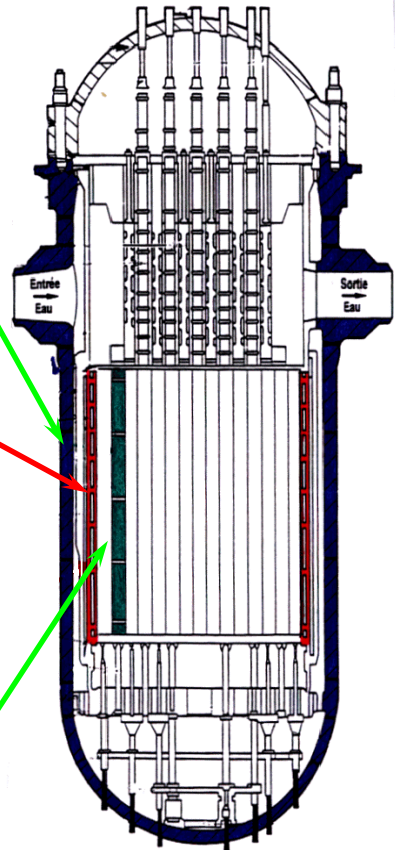
Internals

-barrier

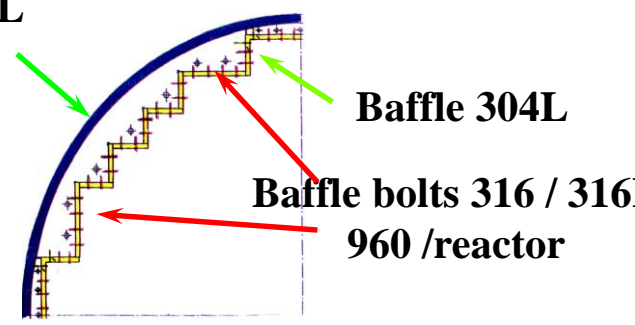
-baffle

-baffle bolts

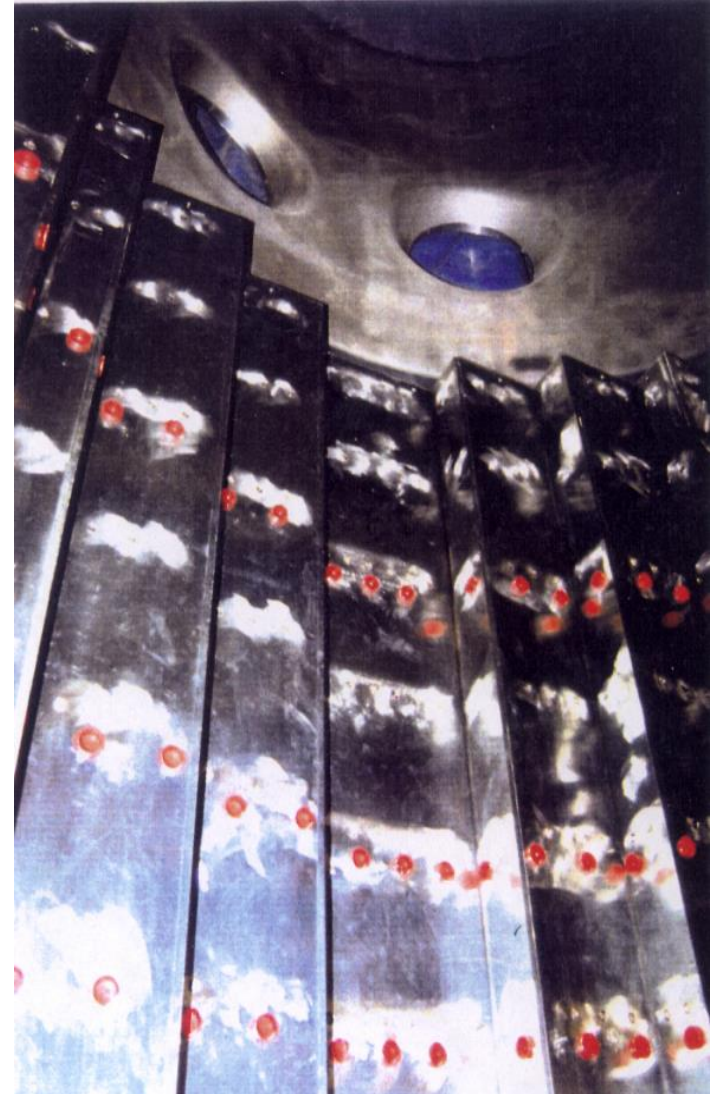
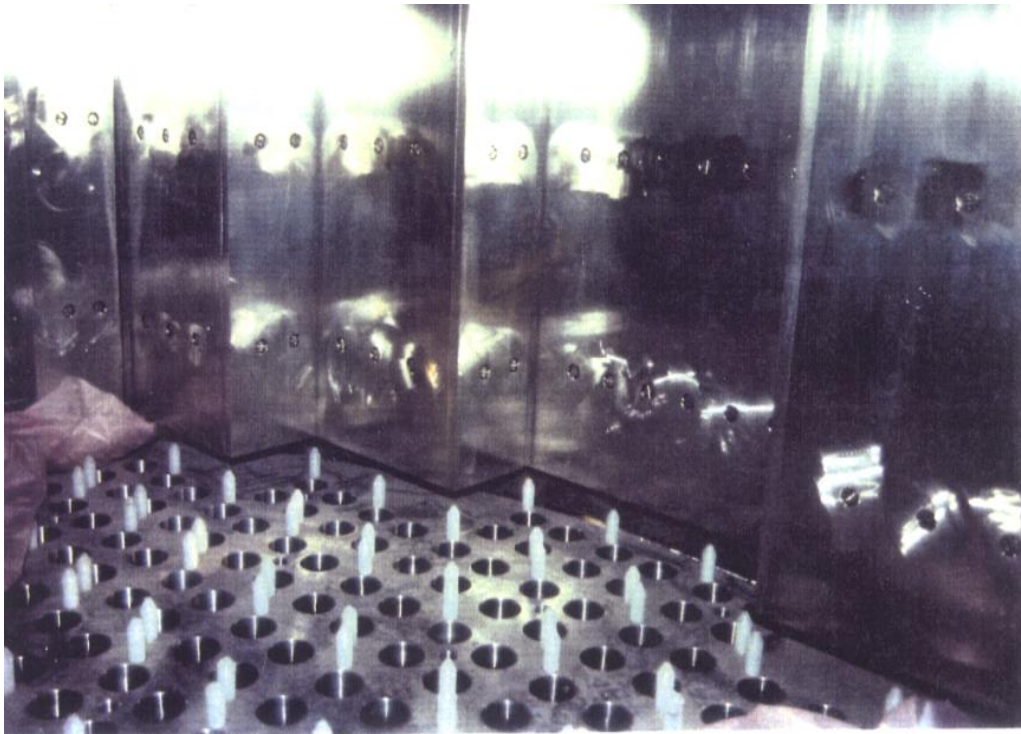
Fuel assembly



Core shroud (envelop) 304L weld 308L



Internals N4 (PWR 1350MW)



core internal structures (core mechanical support, hydraulic, neutronic protection of the vessel)

Baffle

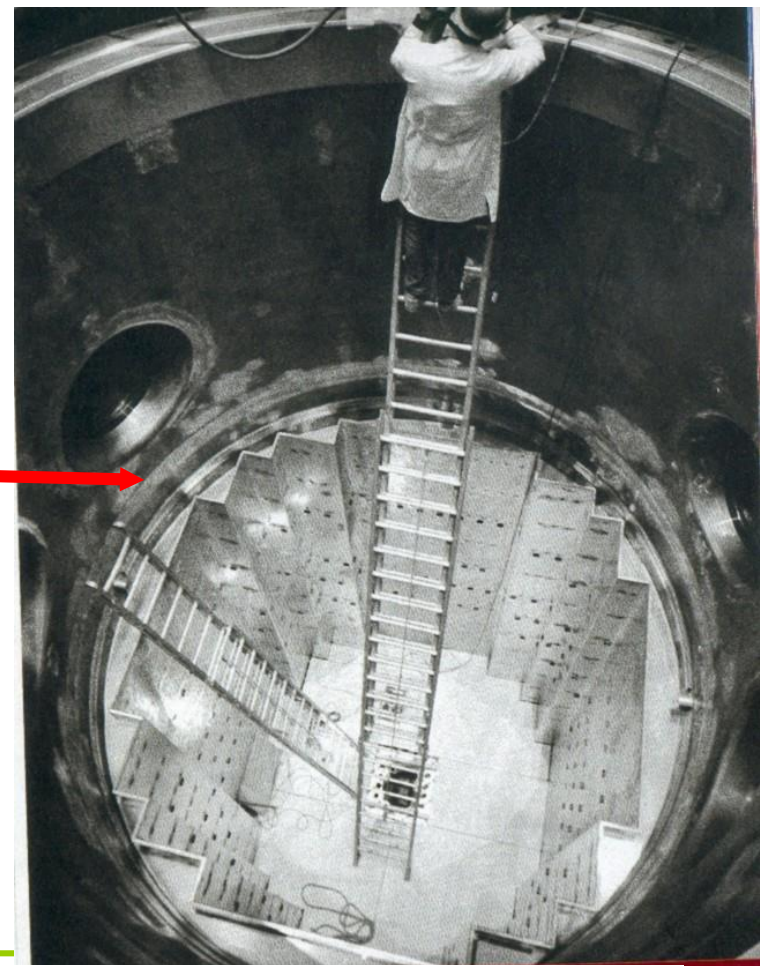
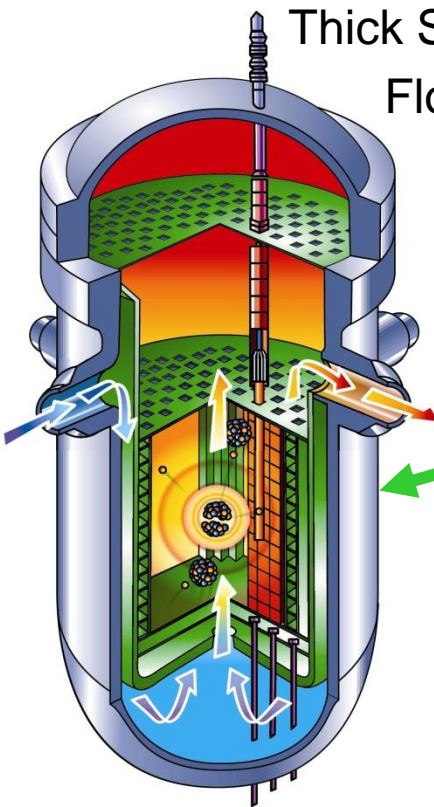
Junction between pressure vessel cylinder and poly-square type fuel assembly core

Thick SS plates screwed together

Flow control

Internal structures of PWR

Internals (temperature, up to 380°C)



18%Cr and 8%Ni (18-08) : 304 et 316

2 to 100 dpa : end of life dose

Irradiation effects: Aggregation of points defaults: loops, voids : **hardening**

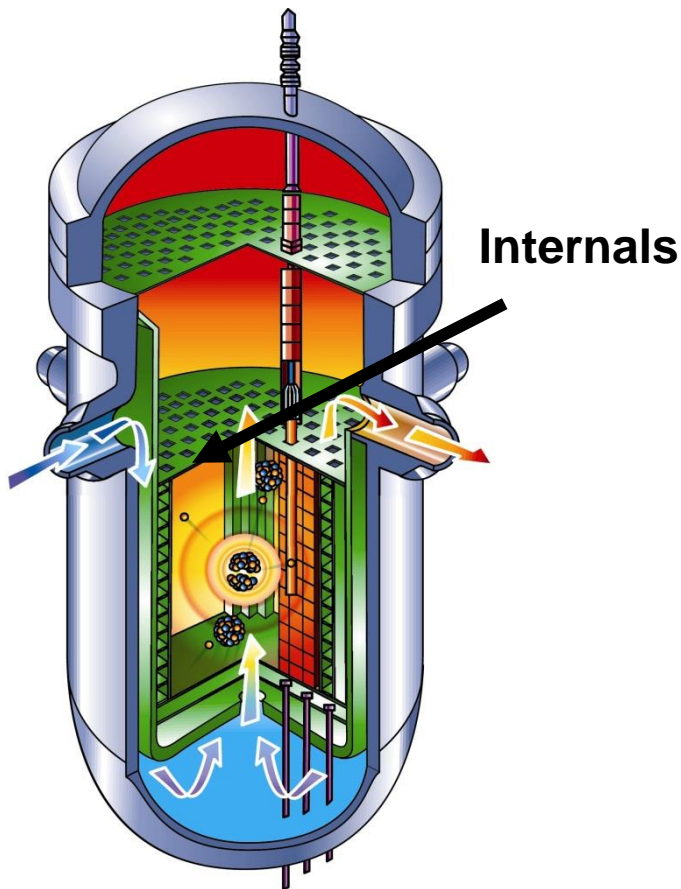
- Depending on the temperature
 Hardening, reduction of ductility,
 Quick increase then saturation

- Swelling is possible at high doses

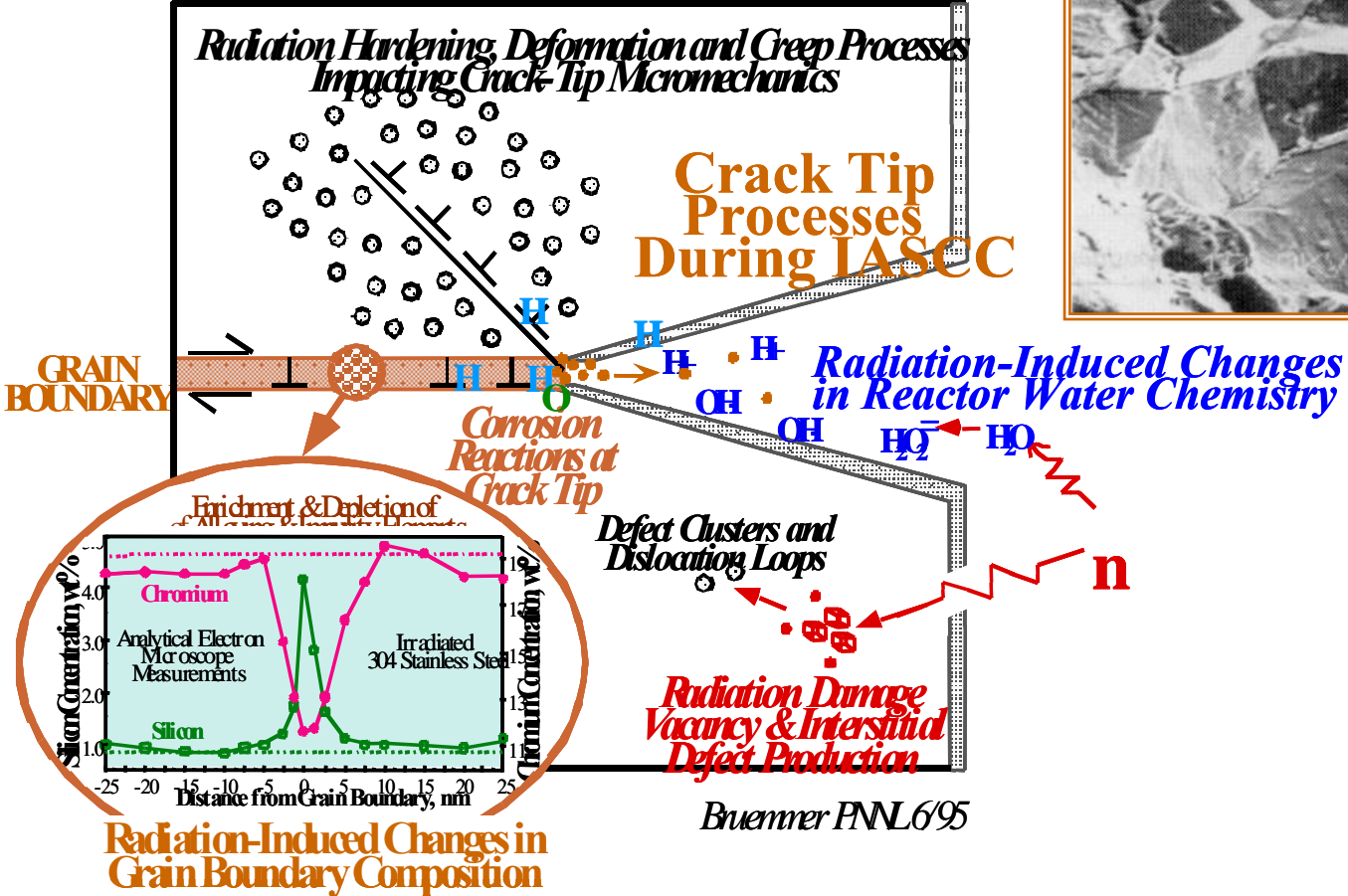
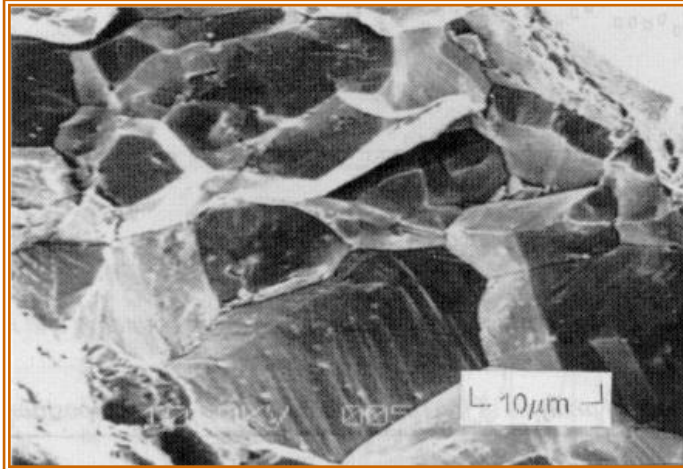
- He formation *in situ* (hardening)

$^{58}\text{Ni} (n,\gamma) ^{59}\text{Ni}$ then $^{59}\text{Ni} (n,\alpha) ^{56}\text{Fe}$

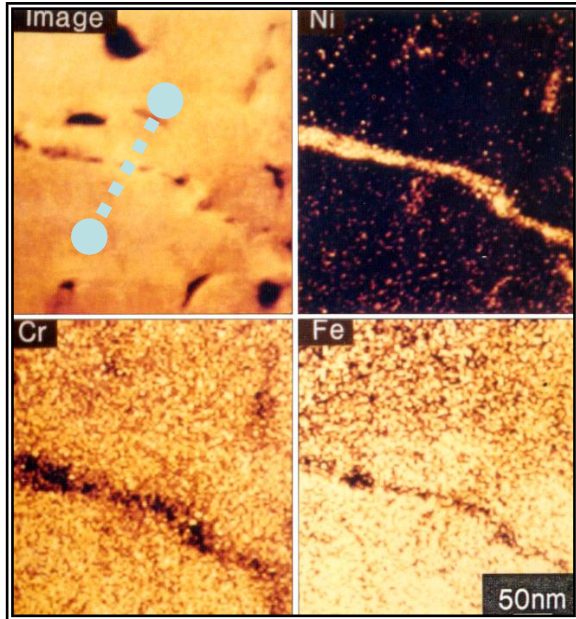
For PWR : 0.5 to 1 ppm He/an



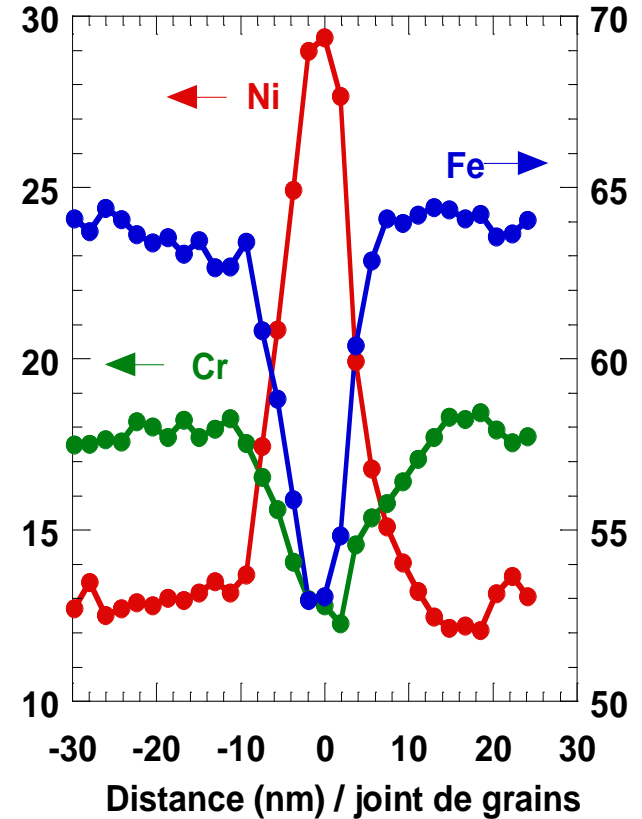
Irradiation assisted stress corrosion cracking



Typical GB chemistry

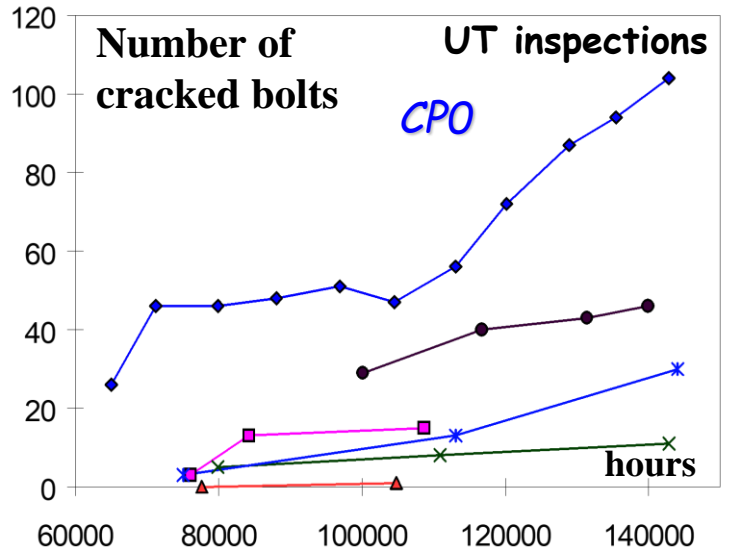
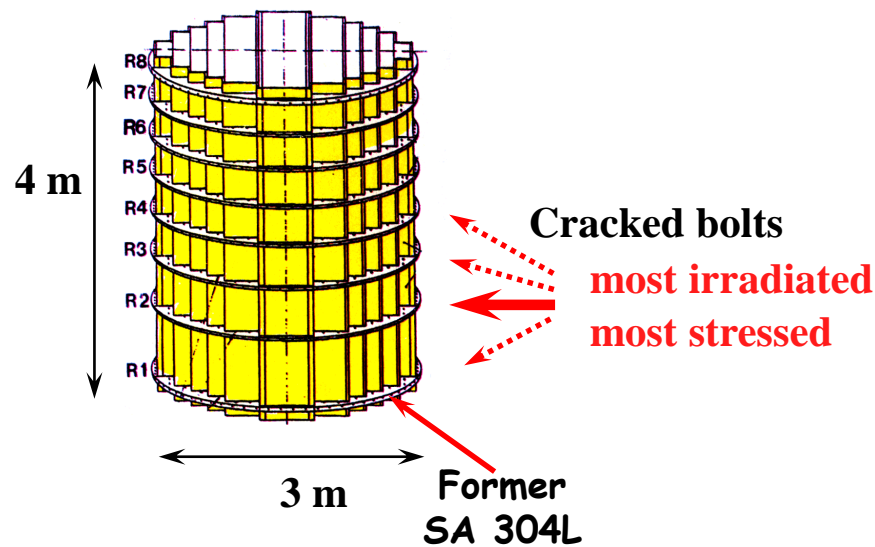


Dose : $3.7 \cdot 10^{21} \text{ n.cm}^{-2}$, i.e. $\approx 0.1 \text{ dpa}$

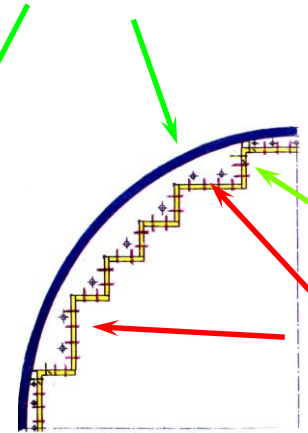


One can observe lowering of Cr content at GB which favor the intergranular rupture. Especially if mechanical loading appears (stress corrosion cracking). This phenomenon is easier when hardening due to irradiation occurs and the corrosive environment which can be confined

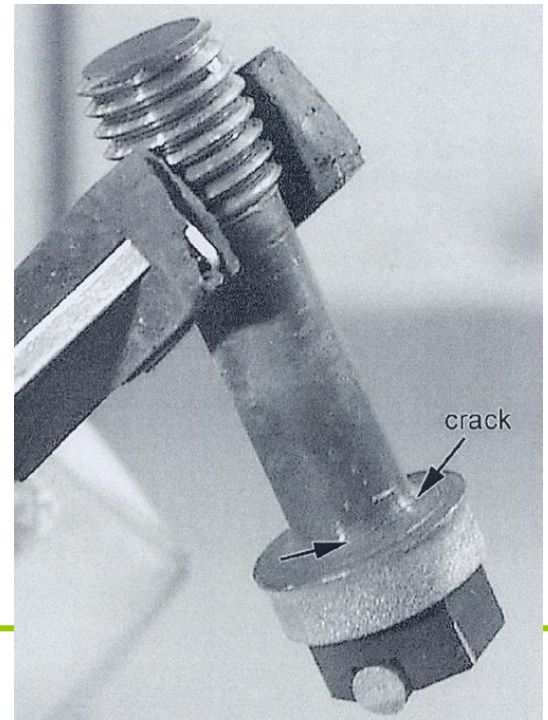
Irradiation effects on Core Internals



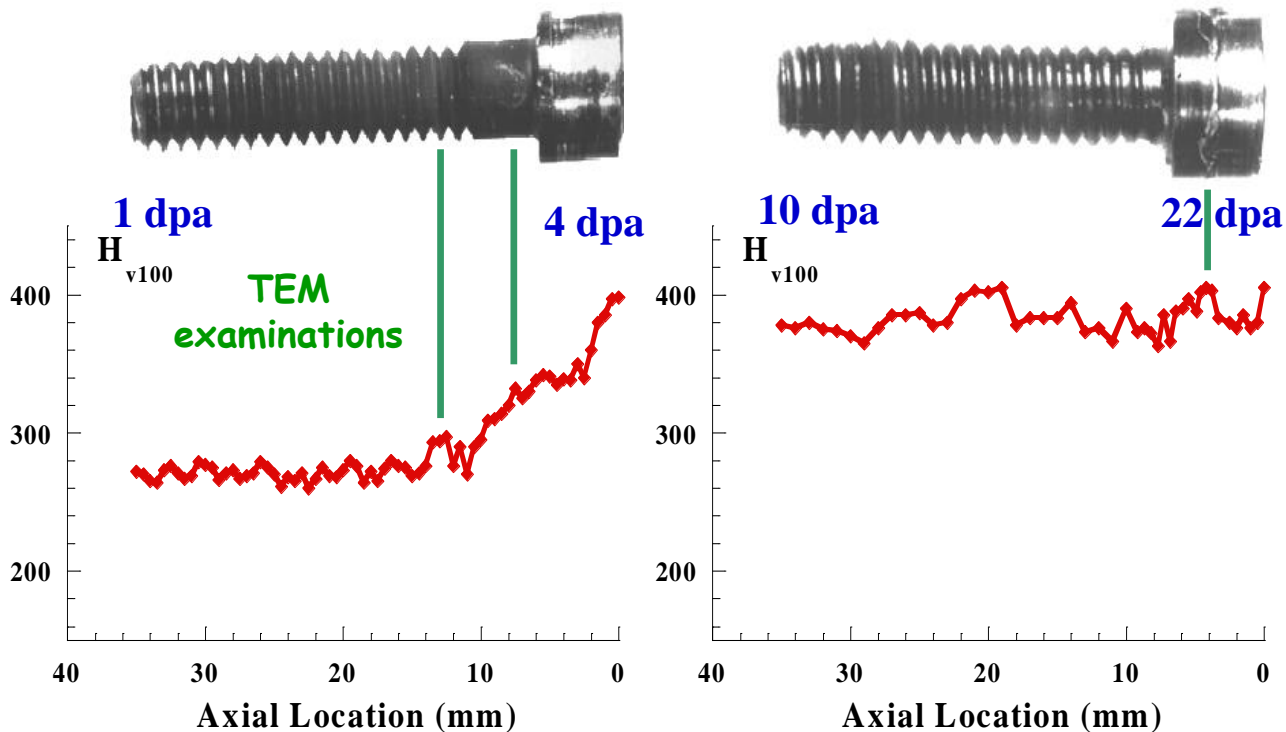
Core barrel 304L
 Welds 308L



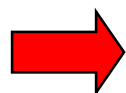
Baffle plate SA 304L
 Baffle bolts CW 316 / 316L
 960 / reactor



Hardness of internal baffle bolts for PWR



-> Saturation of the hardness around 5 dpa

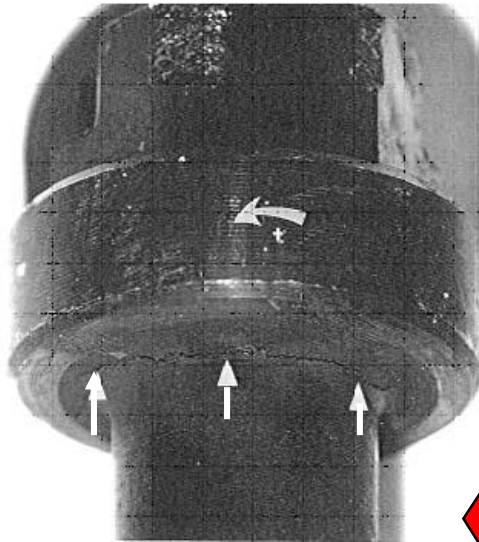


High hardening

Cracking of baffle bolts for PWR internals IASCC

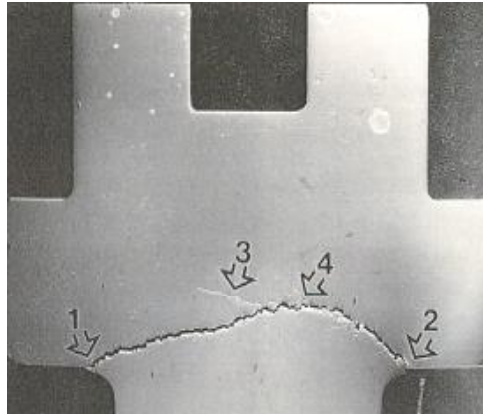
(Irradiation Assisted Stress Corrosion Cracking)

Irradiation dose : 5 - 10 dpa

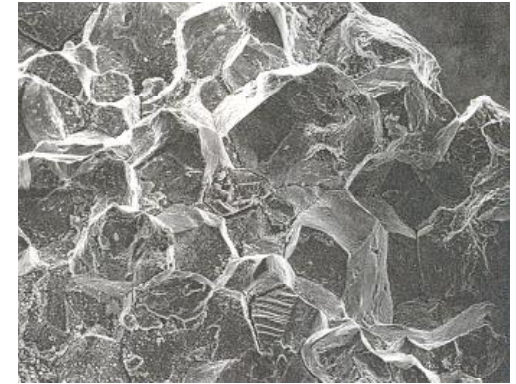


5 mm

Cracking at the junction between: head - deformation area



5 mm



25 μm

Intergranular rupture (out of weld area !)

Stresses, atmosphere(+ irradiation), sensitive material

Solution : reduced loading in the bolts

- Modification of the working conditions of PWR (water flux=>thermohydraulic loading)
- Bolts design: stress concentration
- Replacing some bolts at mid life (at a cost ! To replace 150 bolts, the reactor needs a 3 week outing)
- Thinking about the improvement of the material (for all internals : search for low activation or rapid deactivation materials)